

Heusler Compound/III-V Semiconductor Heterostructures

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Valence Electron Counting

1?

H
2.20

2?

Li	Be
0.98	1.57

Na	Mg
0.93	1.31

K	Ca
0.82	1.00

Rb	Sr
0.82	0.95

Cs	Ba
0.79	0.89

Fr	Ra
0.70	0.90

1?

X_2YZ XYZ

8?

He

3? 4? 5? 6? 7?

B	C	N	O	F	Ne
2.04	2.55	3.04	3.44	3.98	

Al	Si	P	S	Cl	Ar
1.61	1.90	2.19	2.58	3.16	

K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
0.82	1.00	1.36	1.54	1.63	1.66	1.55	1.83	1.88	1.91	1.90	1.65	1.81	2.01	2.18	2.55	2.96	3.00

Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
0.82	0.95	1.22	1.33	1.60	2.16	1.90	2.20	2.28	2.20	1.93	1.69	1.78	1.96	2.05	2.10	2.66	2.60

Cs	Ba	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn
0.79	0.89	1.30	1.50	1.70	1.90	2.20	2.20	2.20	2.40	1.90	1.80	1.80	1.90	2.00	2.20	

Fr	Ra	3?	La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu
0.70	0.90		1.10	1.12	1.13	1.14	1.13	1.17	1.20	1.20	1.10	1.22	1.23	1.24	1.25	1.10	1.27

Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr		
1.10	1.30	1.50	1.70	1.30	1.28	1.13	1.28	1.30	1.30	1.30	1.30	1.30	1.30	1.30	1.30	1.30

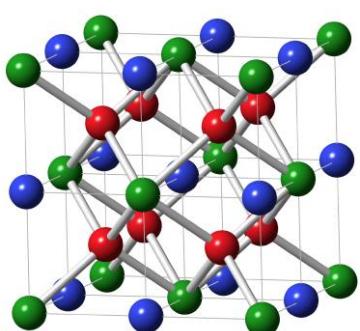
of valence electrons

3? 4? 5? 6? 7? 8? 9? 10? 11? 12?

3?

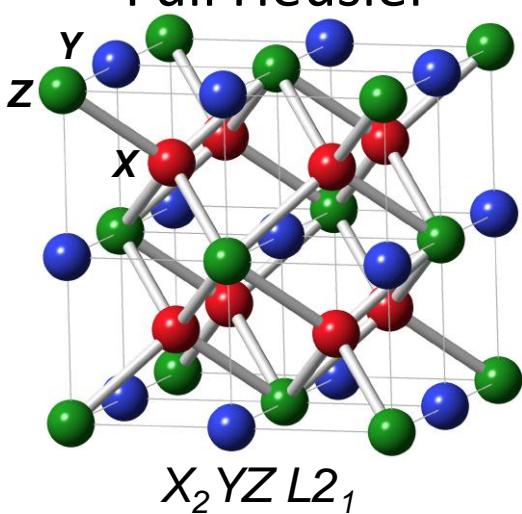
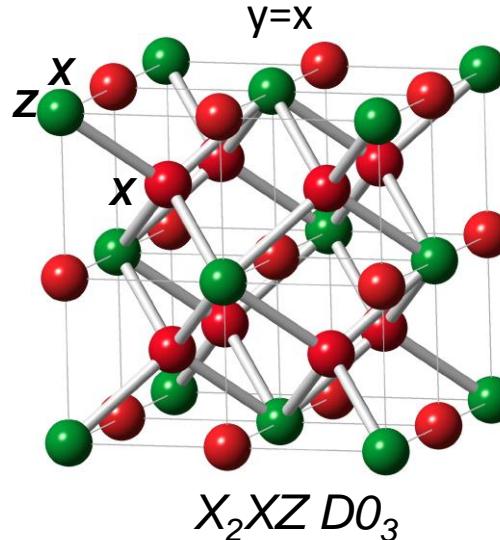
3?

~1000s of combinations!

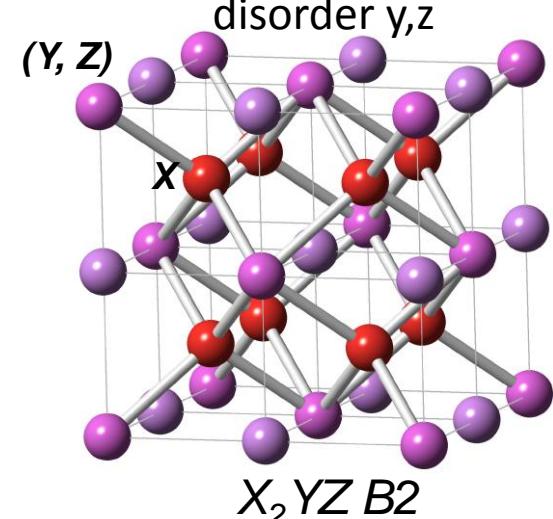


The convergence of research and innovation.

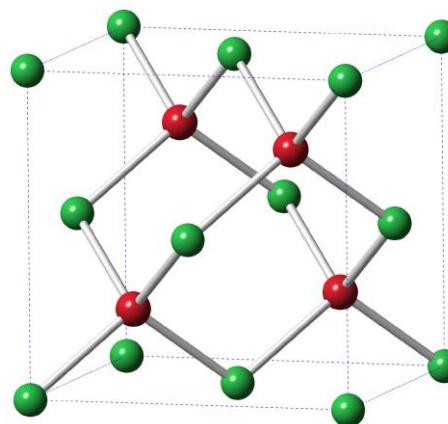
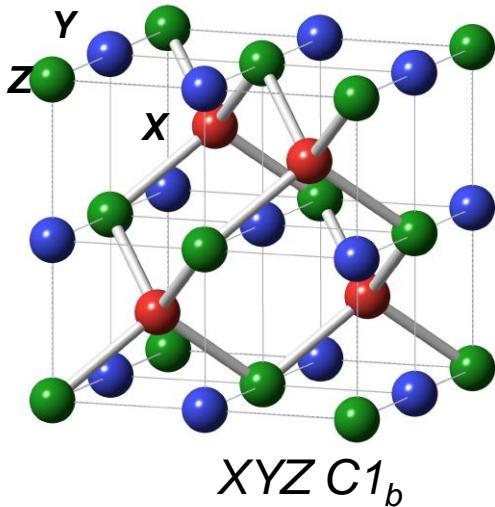
Full Heusler

 $y=x$ 

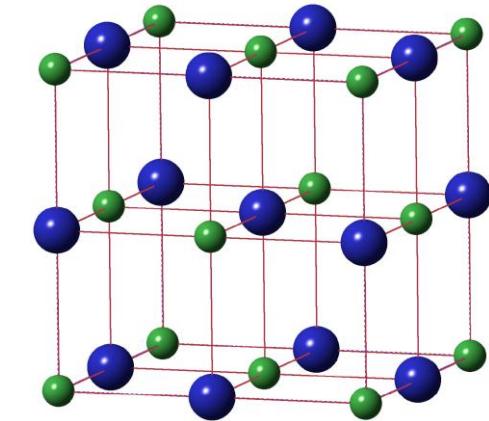
disorder y,z



Half Heusler

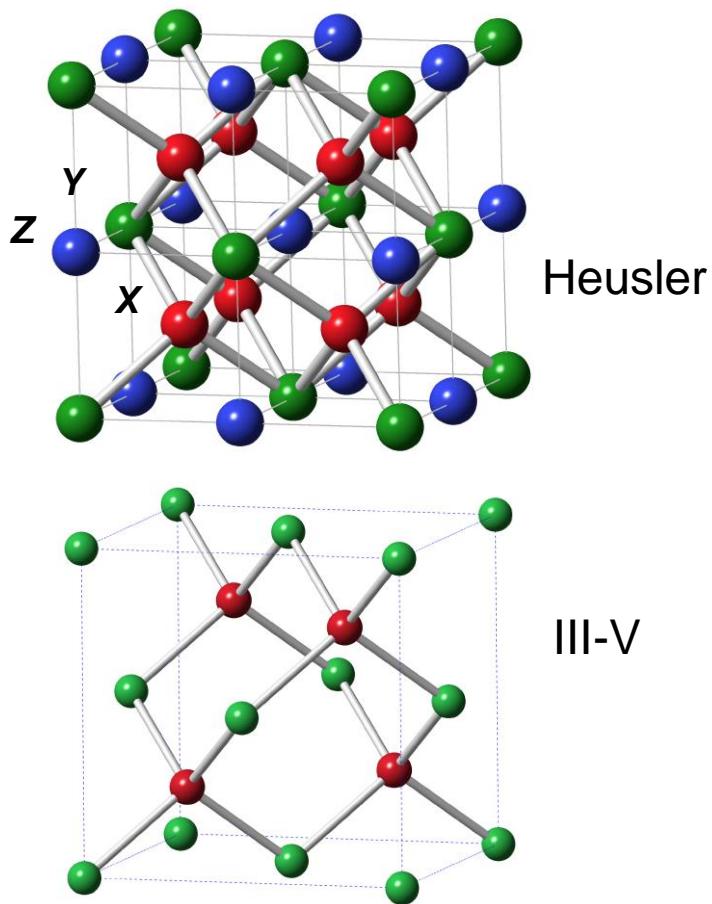


GaAs

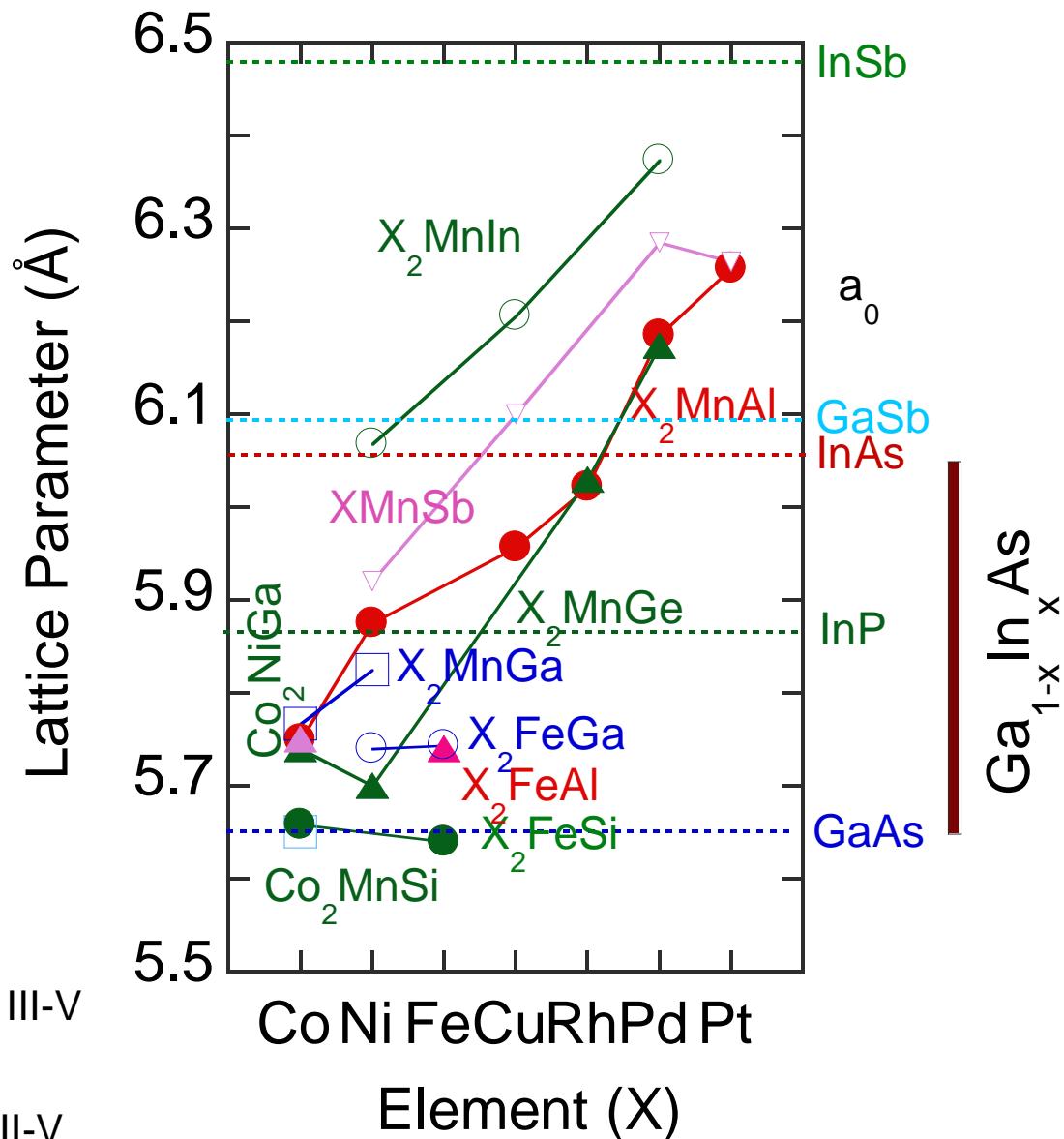


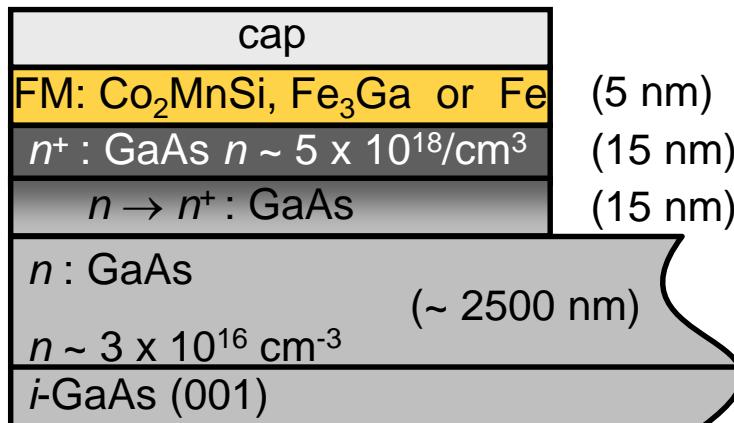
ErAs, MgO

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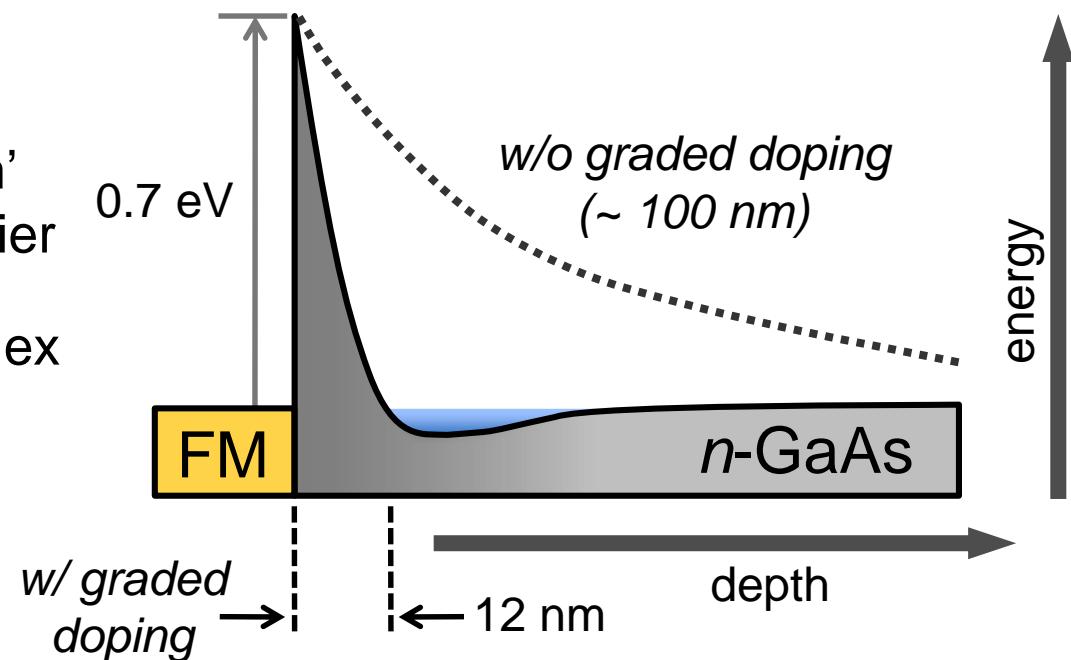
- High spin polarization
- Closely lattice matched to the III-V semiconductor
- Thermodynamic stability on III-V semiconductor

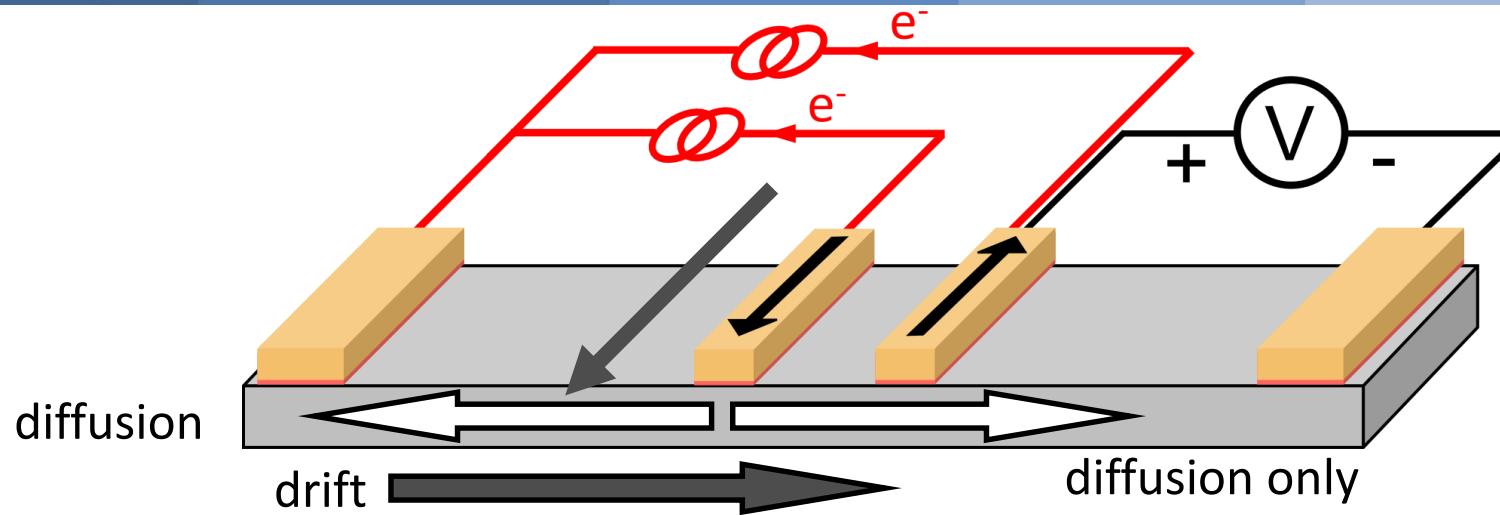




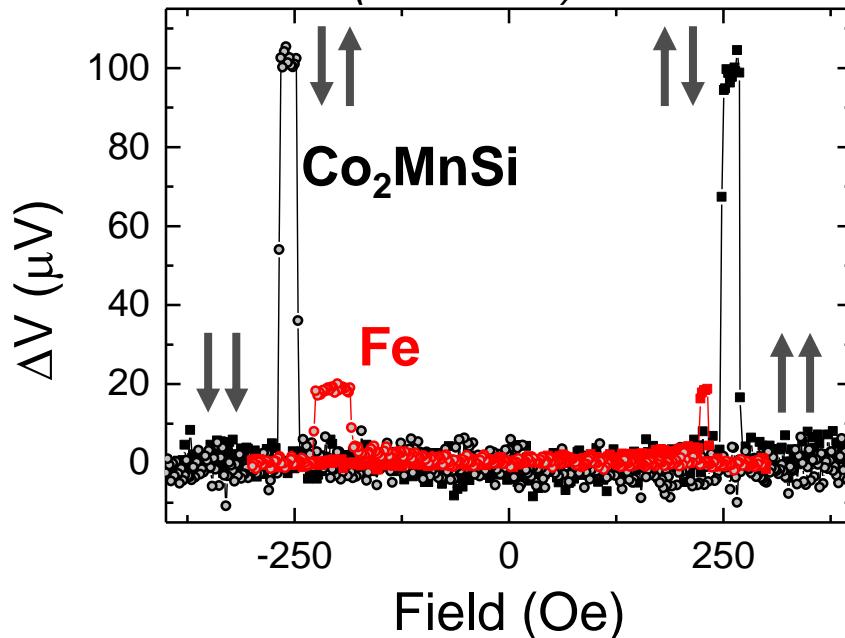
- Epitaxially grown along [001]
- Fe polarization at Fermi level
- Co₂MnSi proposed to be half-metallic
- Surface-induced FM anisotropy

- Graded doping used to ‘thin’ natural forming Schottky barrier
- Interface states lead to complex bias dependence

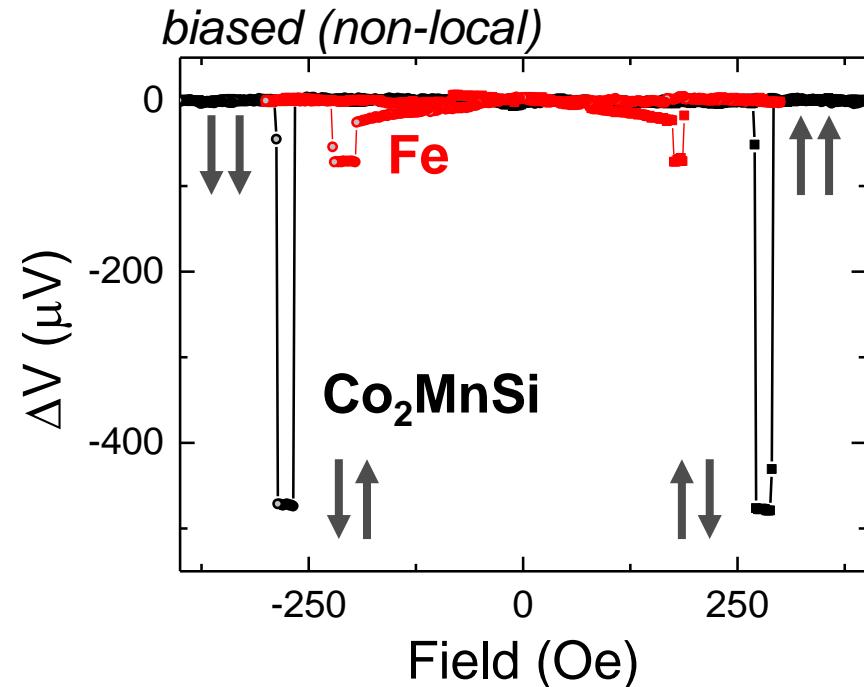




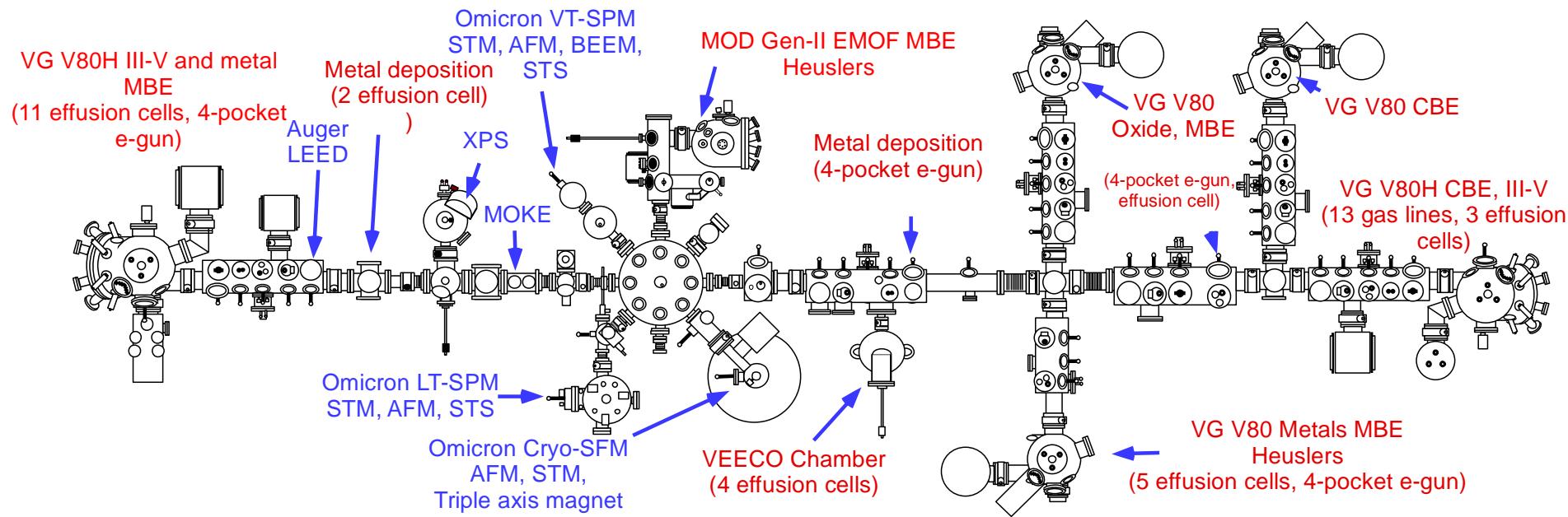
unbiased (non-local)



biased (non-local)



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in-situ growth and atomic level characterization

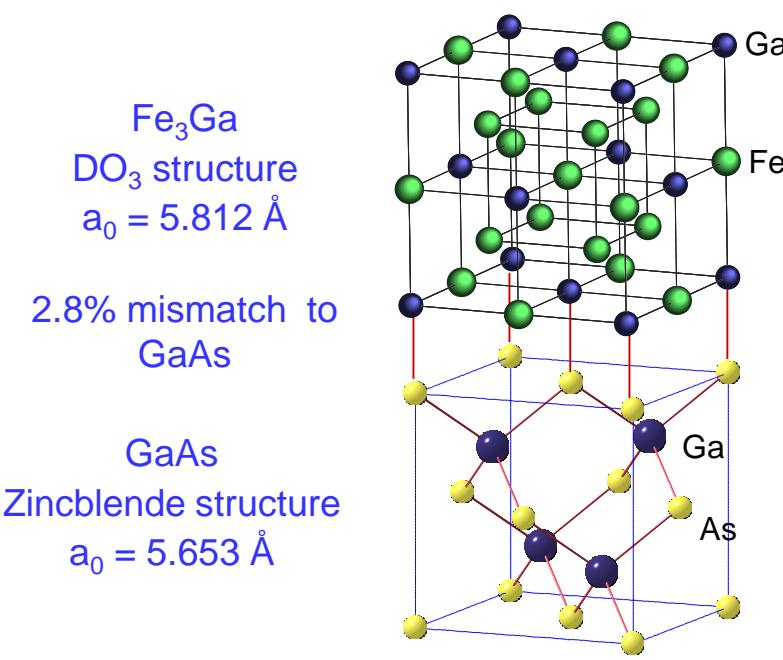
Enhanced growth capabilities interconnected MBE/CBE systems for III-Vs, metals, metallic compounds and oxides

Determination of structure and chemistry at the atomic level at different stages of growth
STM/AFM, Auger, XPS, LEED, RHEED, MOKE

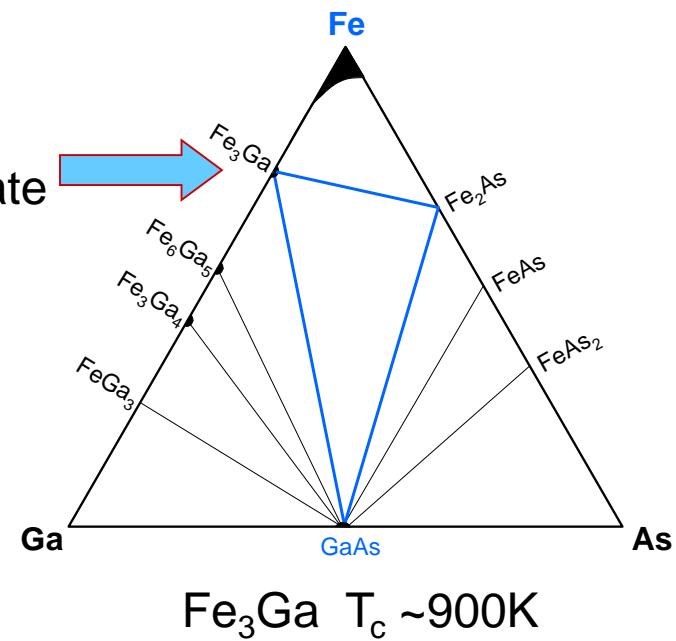
Atomic level electronic and magnetic properties – STM/STS, BEEM (VTSTM 50-800K),
LT-SPM (4-300K), Cryo-SFM (~4-300K)

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stable structure (Pearson)	Phase	Space Group	Structure	Type	Lattice Parameter
	$\alpha\text{-Fe}_3\text{Ga}$	Pm $\bar{3}m$	L ₁ ₂	Cu ₃ Au	$a = 3.678 \text{ \AA}$
	$\alpha'\text{-Fe}_3\text{Ga}$	Pm $\bar{3}m$	B2	CsCl	$a = 2.91 \text{ \AA}$
	$\alpha''\text{-Fe}_3\text{Ga}$	Fm $\bar{3}m$	D0 ₃	BiF ₃	$a = 5.808 \text{ \AA}$
	$\beta\text{-Fe}_3\text{Ga}$	P6 ₃ /mmc	D0 ₁₉	Ni ₃ Sn	$a = ?, c = ?$

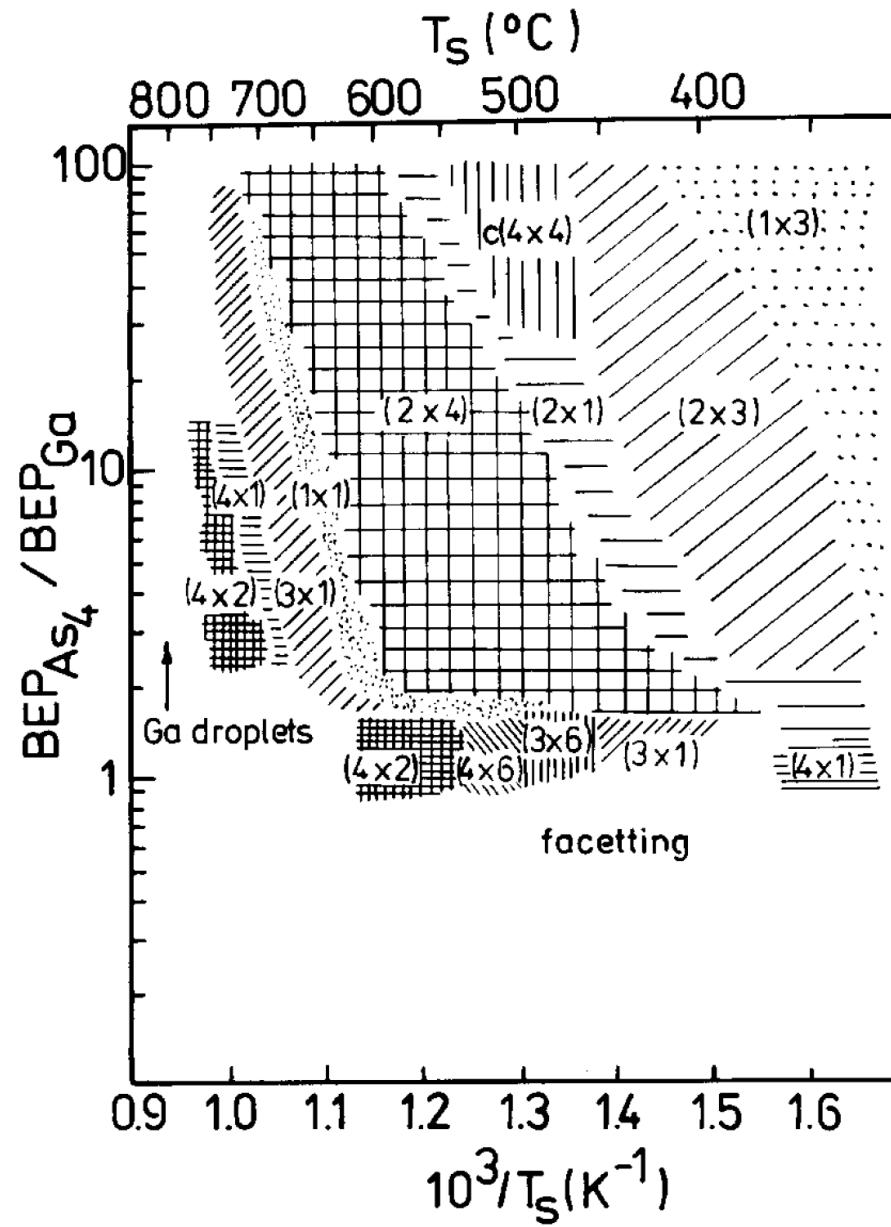


Fe₃Ga is a
good candidate



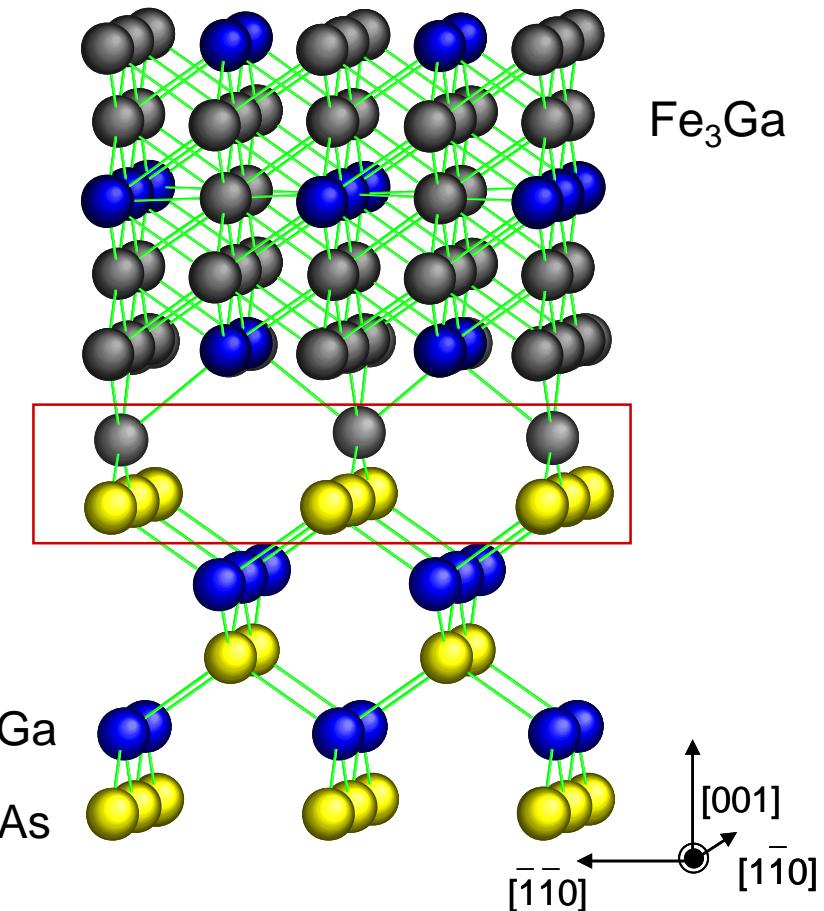
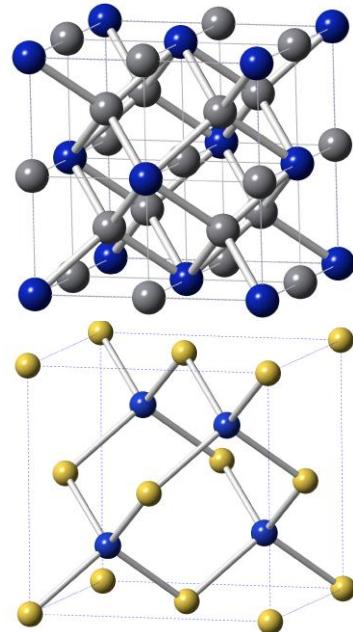
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GaAs(001) has multiple surface reconstructions depending on As/Ga surface composition



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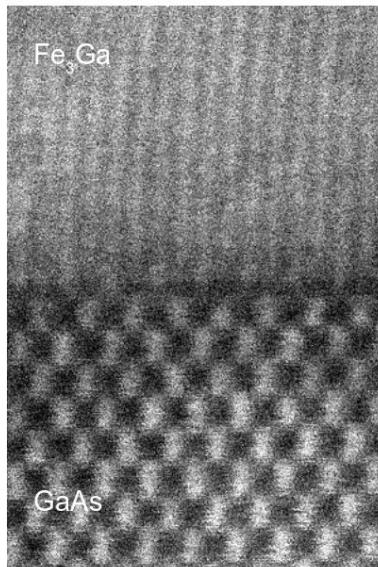
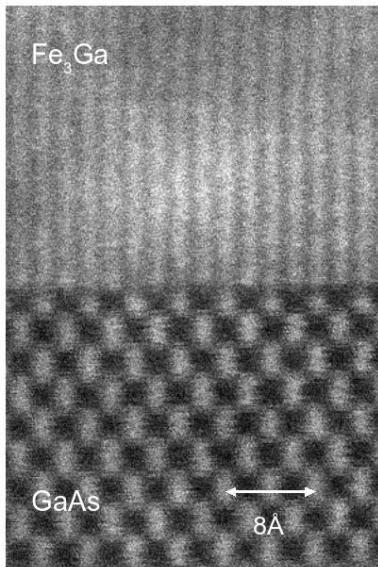
- **GaAs surface reconstruction**
- As-rich
- Ga-rich
- how As- or Ga- rich, multiple reconstructions are possible
- **Initiation of Fe_3Ga growth**
- Fe first
- Ga first
- Co-deposition of Fe+Ga



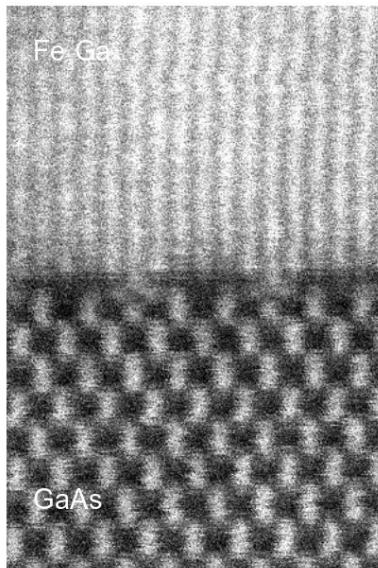
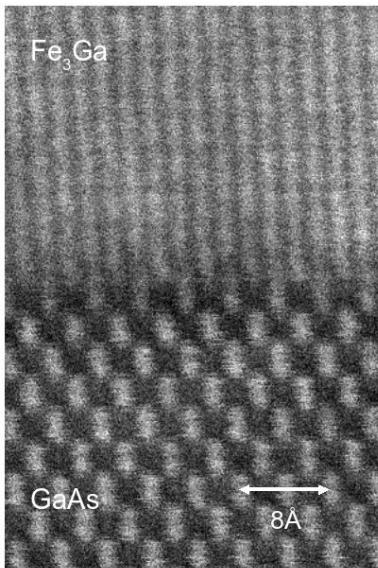
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HAADF-STEM images

As-rich



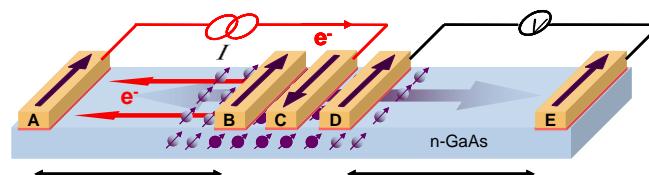
Ga-rich



[1-10]

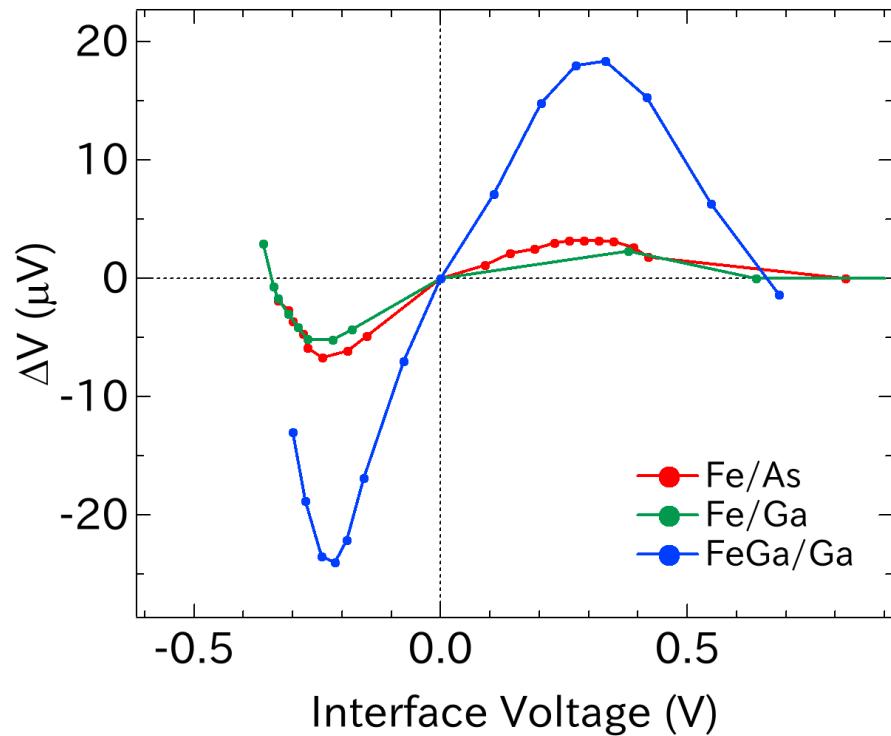
[110]

Bias dependence of the non-local signal



Forward Bias

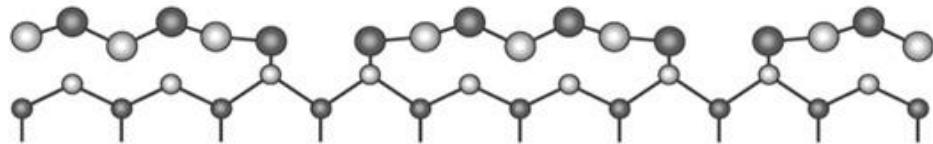
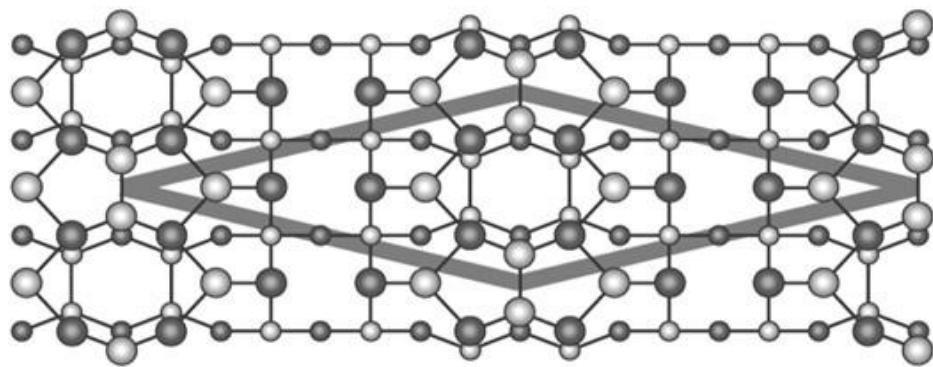
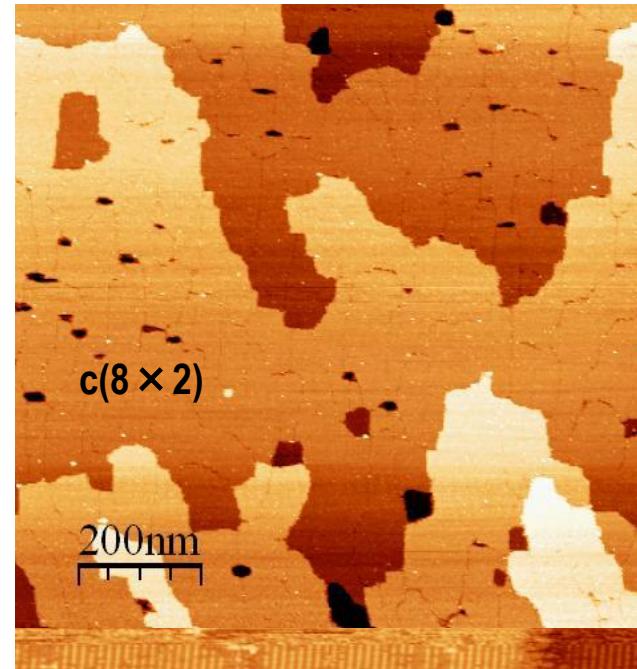
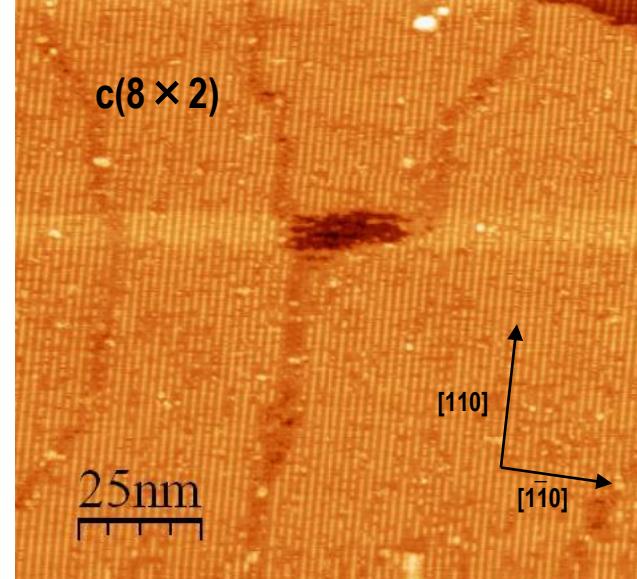
Reverse Bias

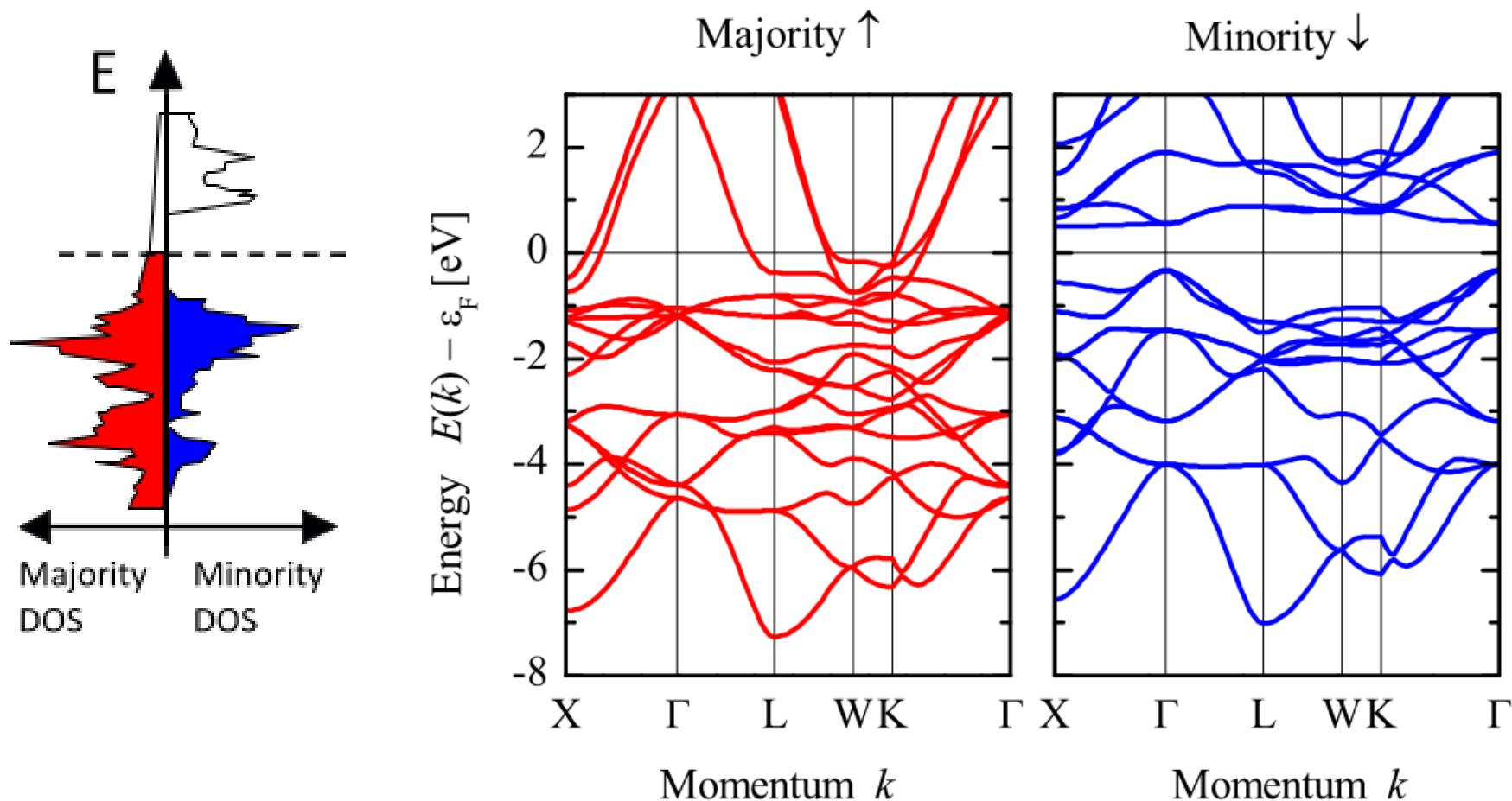


Interface Voltage (V)

- Different atomic structures for As- and Ga-rich interface
- Distinct magnitude and bias dependence of Spin-Valve signal

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 ζ model for $c(8 \times 2)$ $[1\bar{1}0]$
 $[110]$ \bullet : Ga \bullet : As $[001]$
 $[110]$  $c(8 \times 2)$ 25nm $[110]$
 $[1\bar{1}0]$ 



- Large minority spin gap of approximately 1eV is centred around the Fermi level
Co₂MnSi – 0.06% mismatch to GaAs – not thermodynamically stable on GaAs

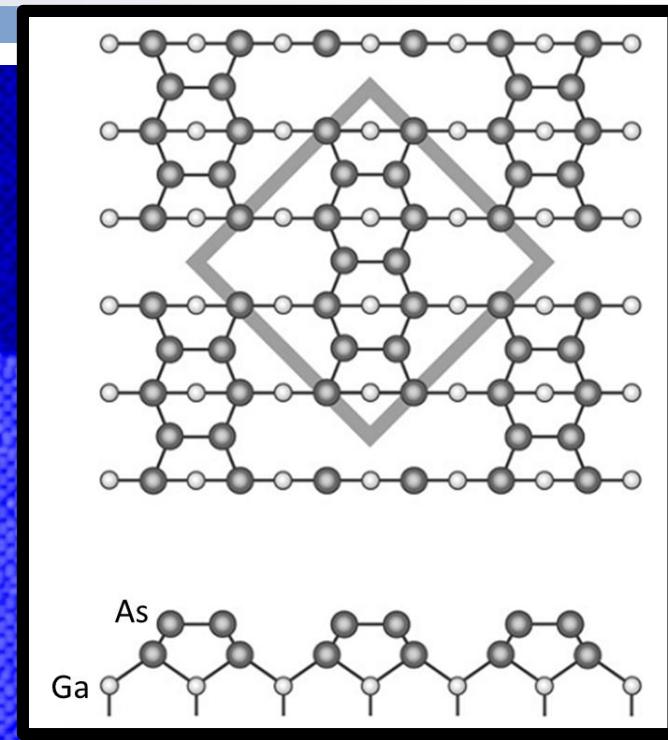
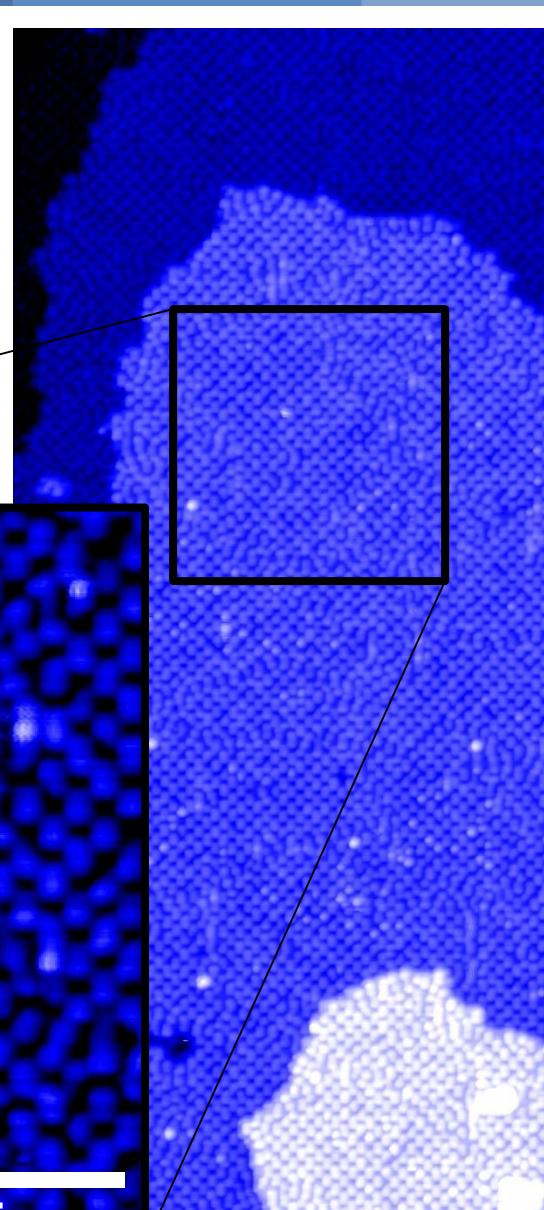
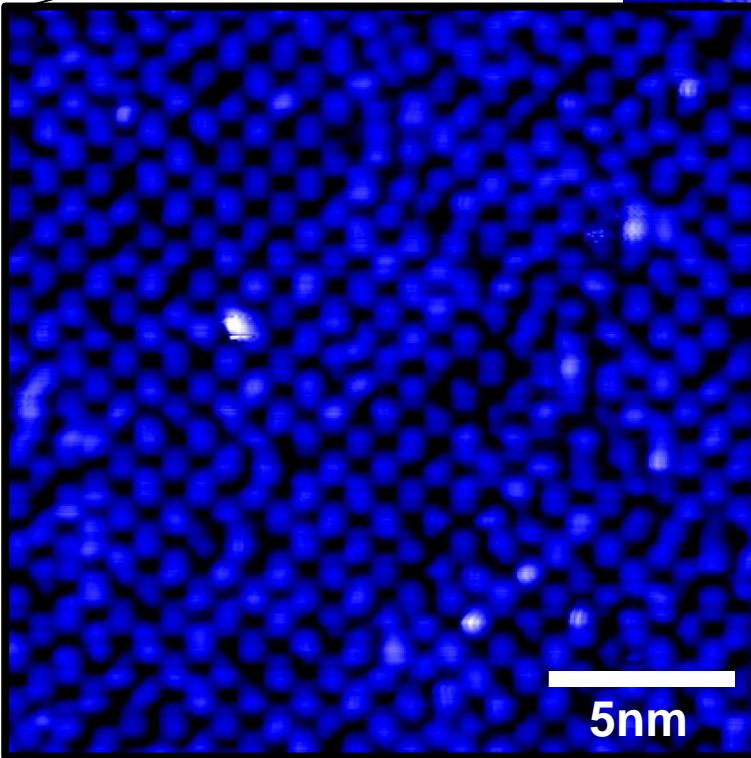
Balke, B., S. Ouardi, et al. (2010). Solid State Communications **150**(11–12): 529-532.

M. Jourdan et al., Nature Communications, **5** 3974 (2014)

GaAs(001) c(4x4) As-rich surface

Extra $\frac{3}{4}$ ML of As

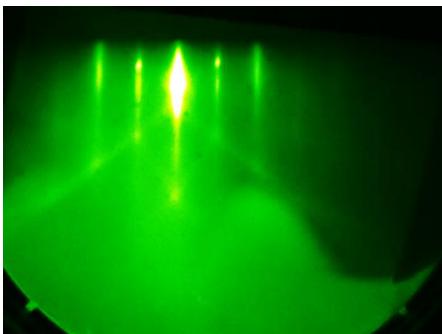
Filled States image



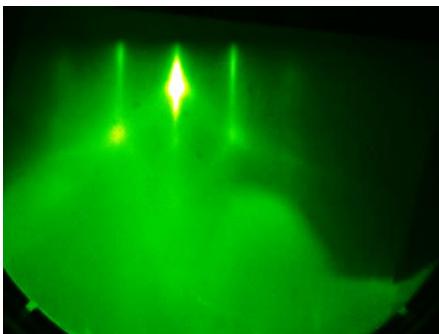
Growth of Co_2MnSi on GaAs(001)-c(4x4)

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[110]



[010]

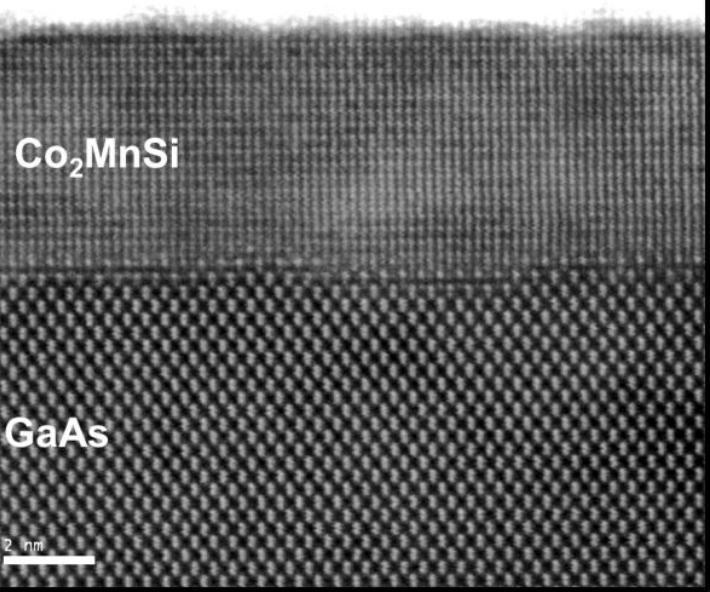


Growth Temperature $\sim 270^\circ \text{ C}$

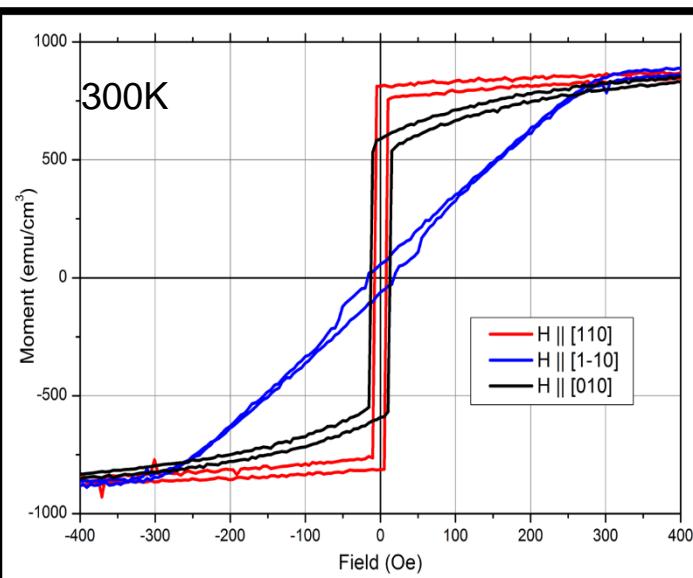
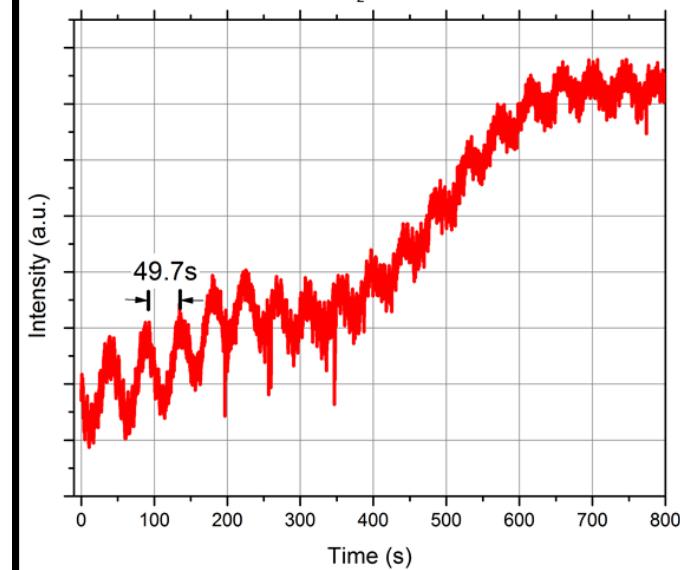
HAADF-STEM of CMS/GaAs

Heterostructure

Al/Au

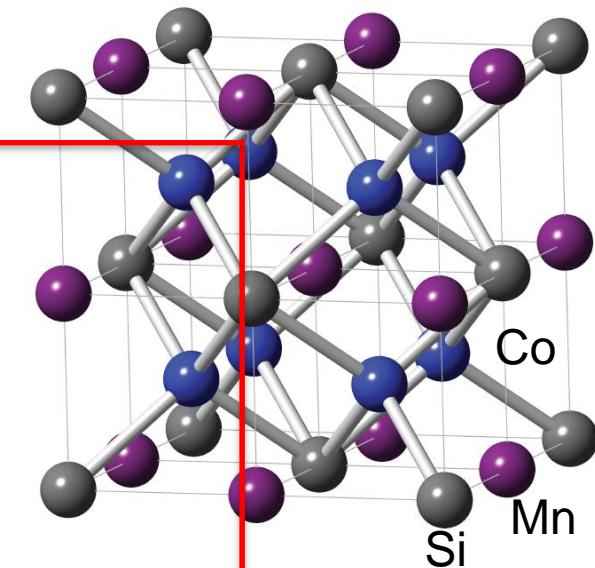
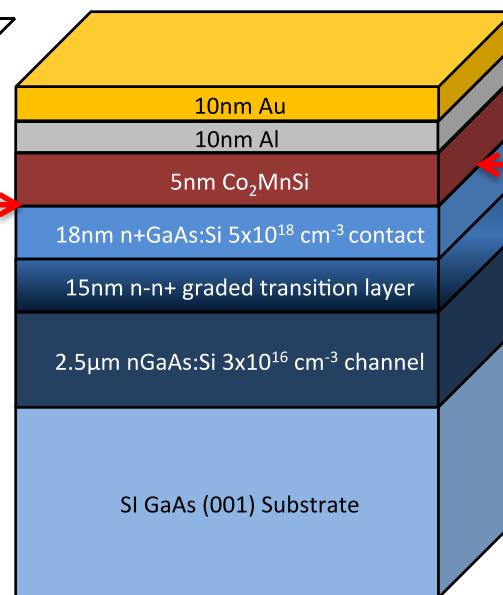
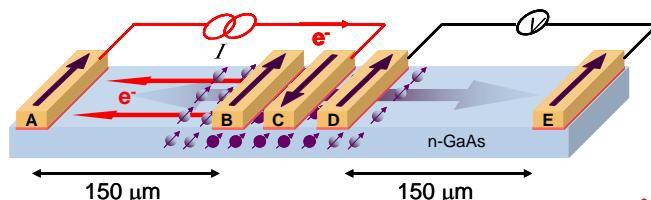


RHEED Oscillations of Co_2MnSi Layer by Layer Growth

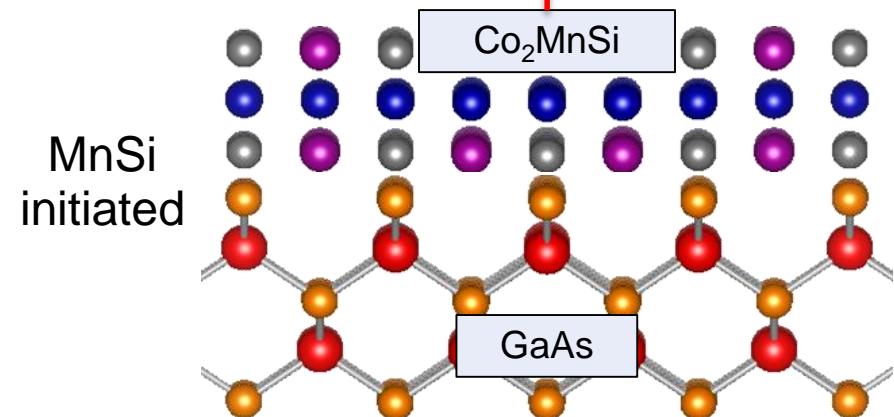
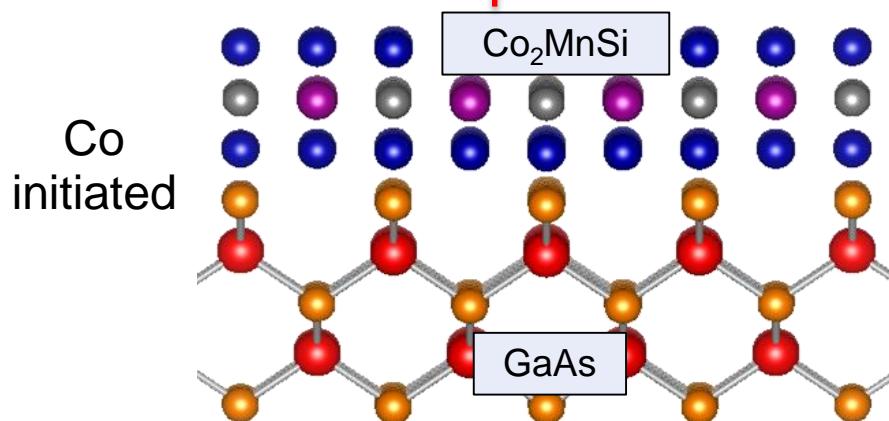


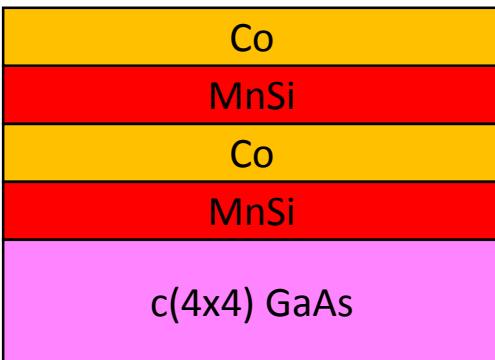
Co₂MnSi/GaAs Spin Contacts

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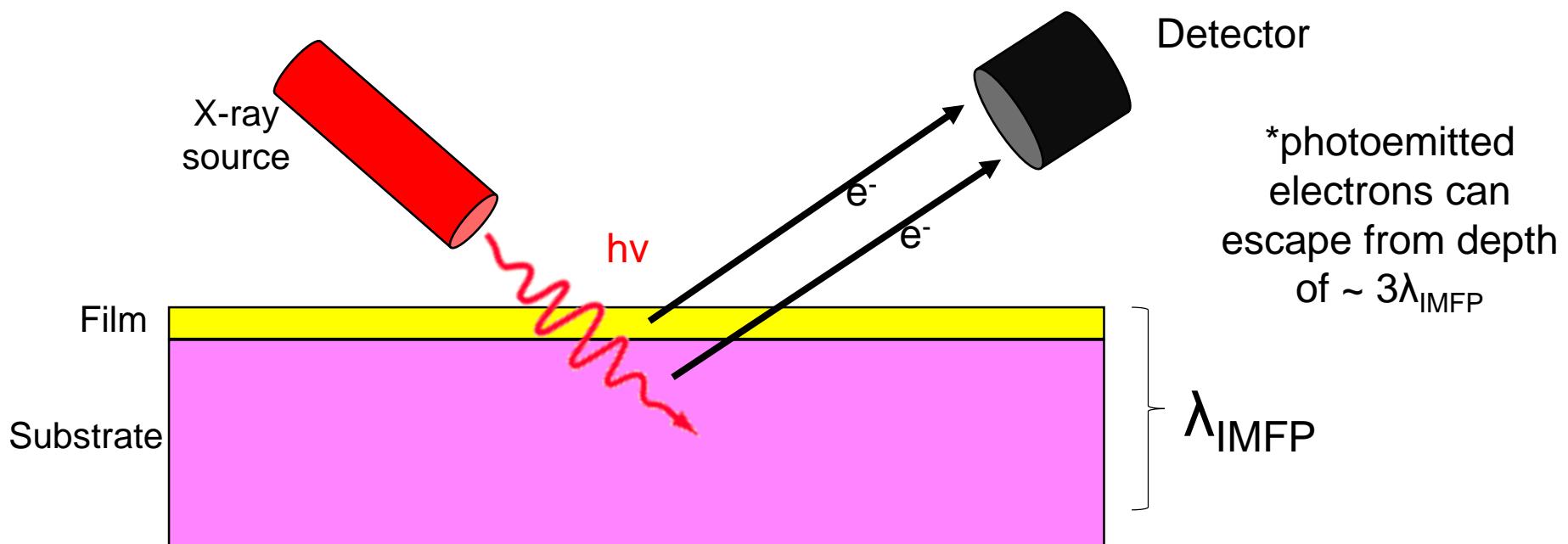


T_g 270° C

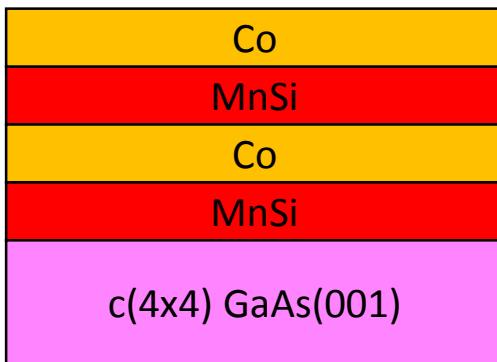


Grow Co₂MnSi seed layer using two different nucleation sequences**MnSi initiated growth****Co initiated growth**

X-ray photoemission spectroscopy (XPS) allows study of core level intensity as a function of film thickness



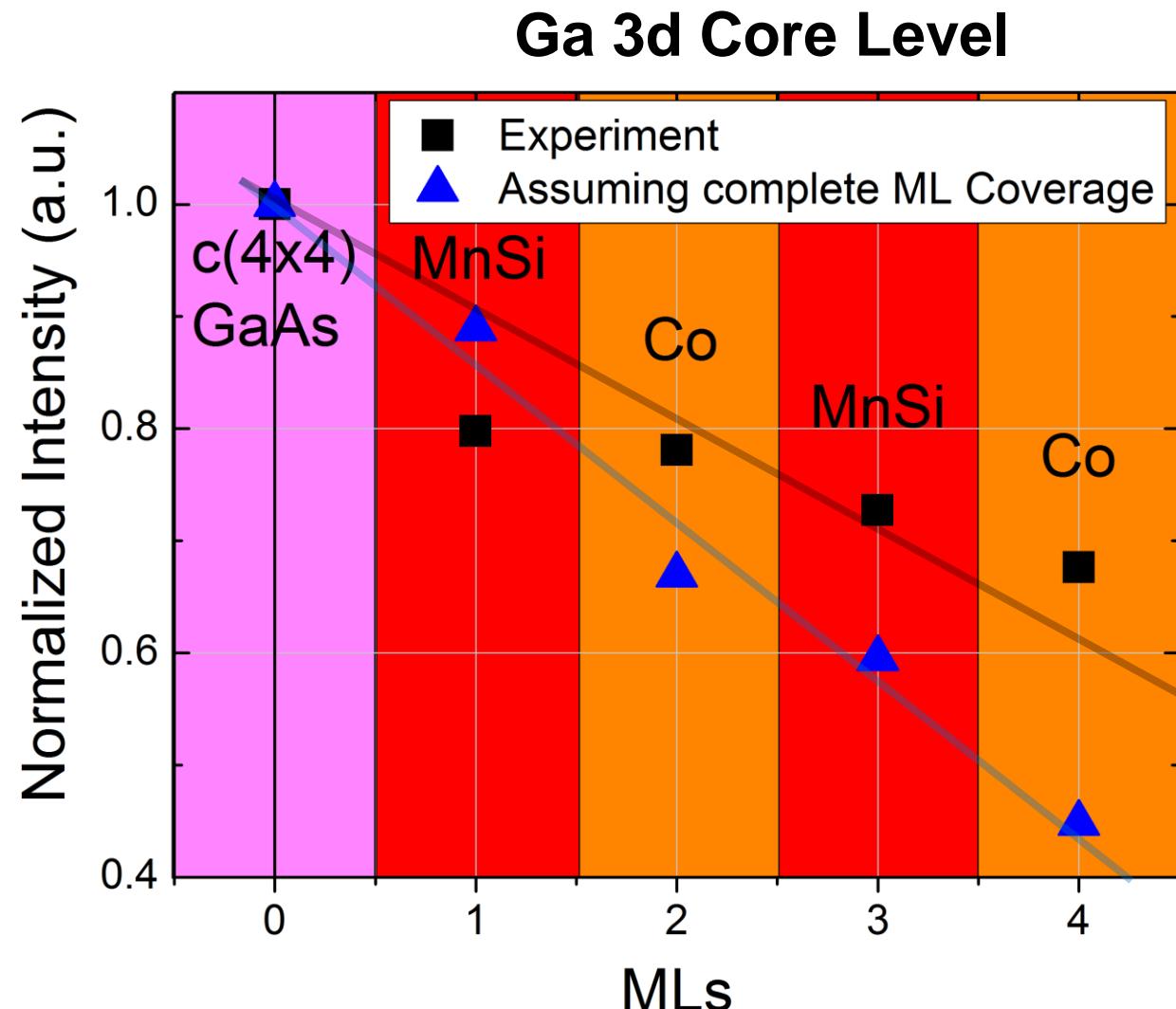
MnSi initiated growth



As and Ga 3d core levels show similar decreases in intensity for each layer deposited

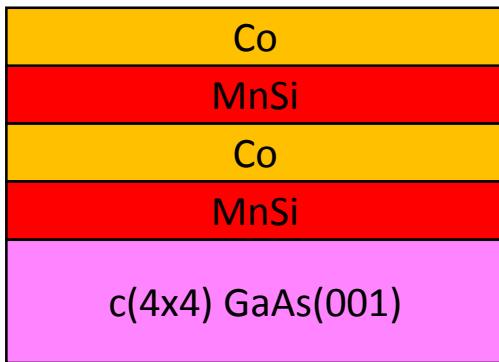
As and Ga intensity does not attenuate as fast as expected for simple ML by ML coverage on GaAs

Suggests that Ga and As must be riding on the surface or island growth



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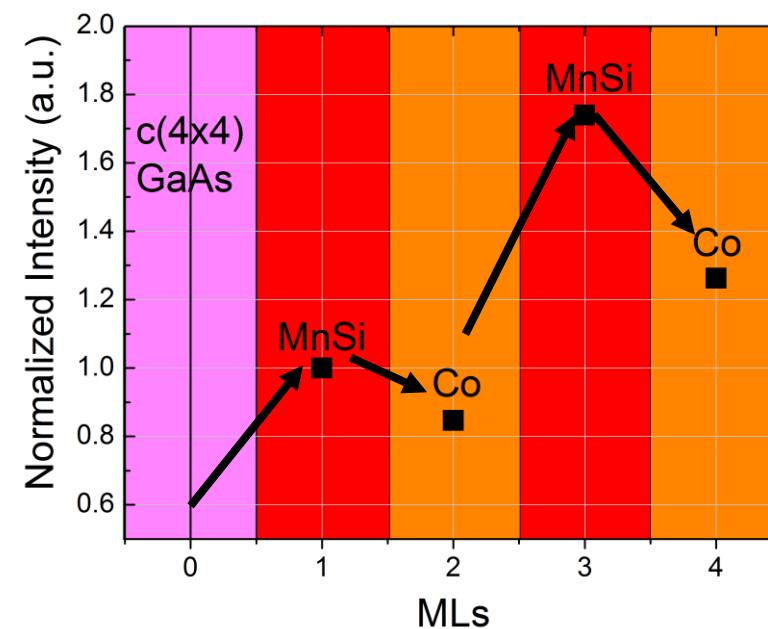
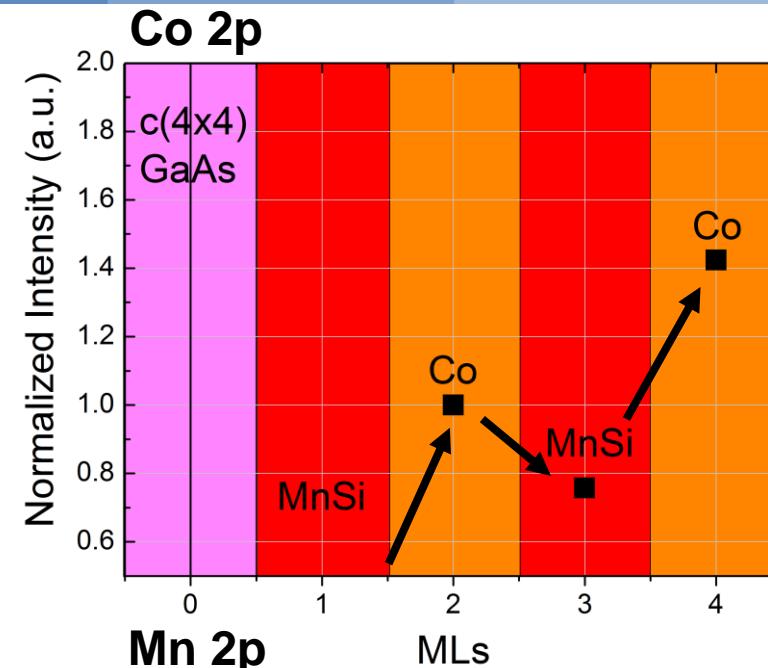
MnSi initiated growth



MnSi deposition attenuates Co 2p peak

Co deposition attenuates Mn 2p peak

MnSi layers cover Co layers and Co layers cover MnSi layers (simple layer-by-layer growth)



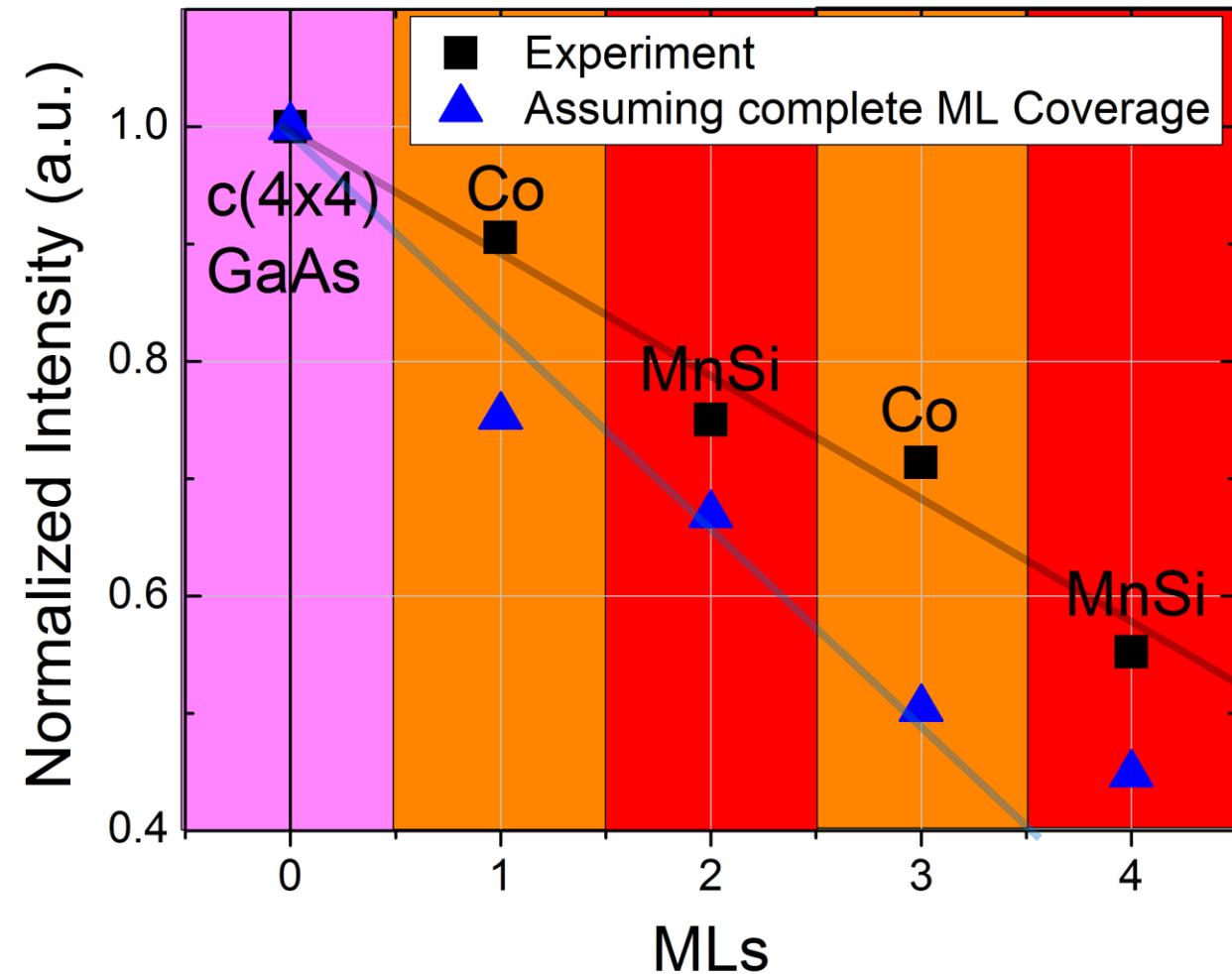
Co initiated growth



As and Ga 3d core levels show similar decreases in intensity for each layer deposited

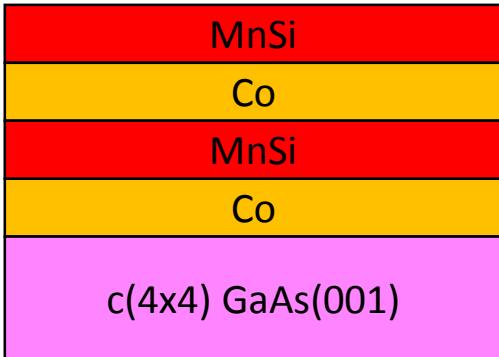
Similar to MnSi initiated growth, Ga and As core levels do not attenuate as fast as expected for simple layer-by-layer coverage, implying **Ga and As ride on surface during growth**

Ga 3d Core Level



1ML Co Initiated Growth

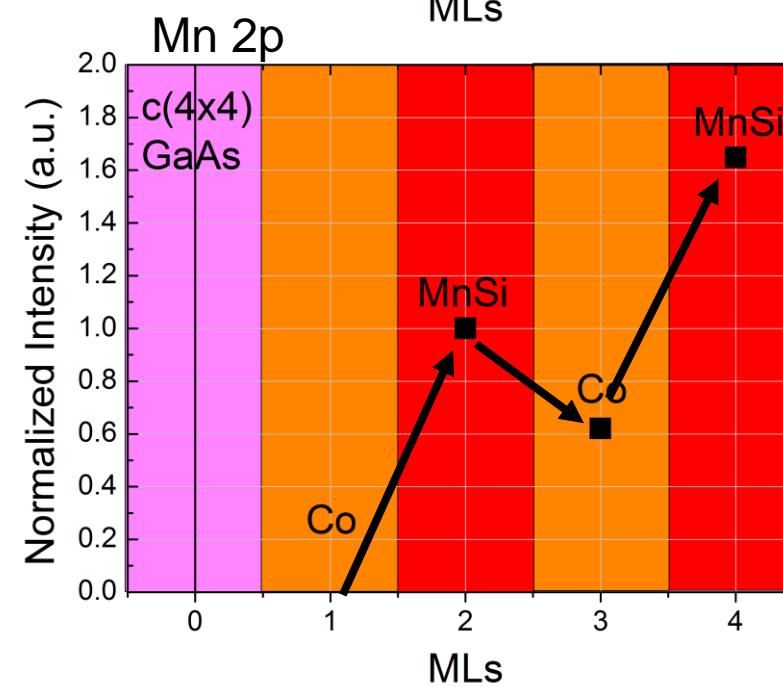
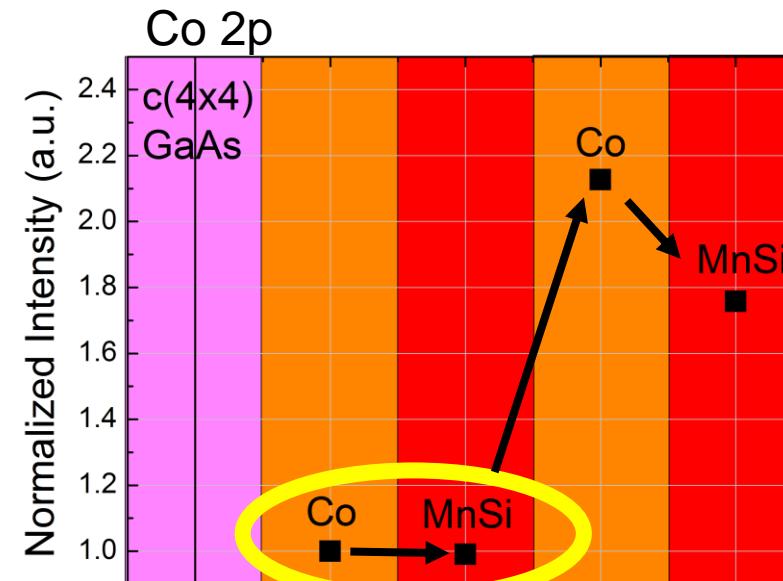
Co initiated growth



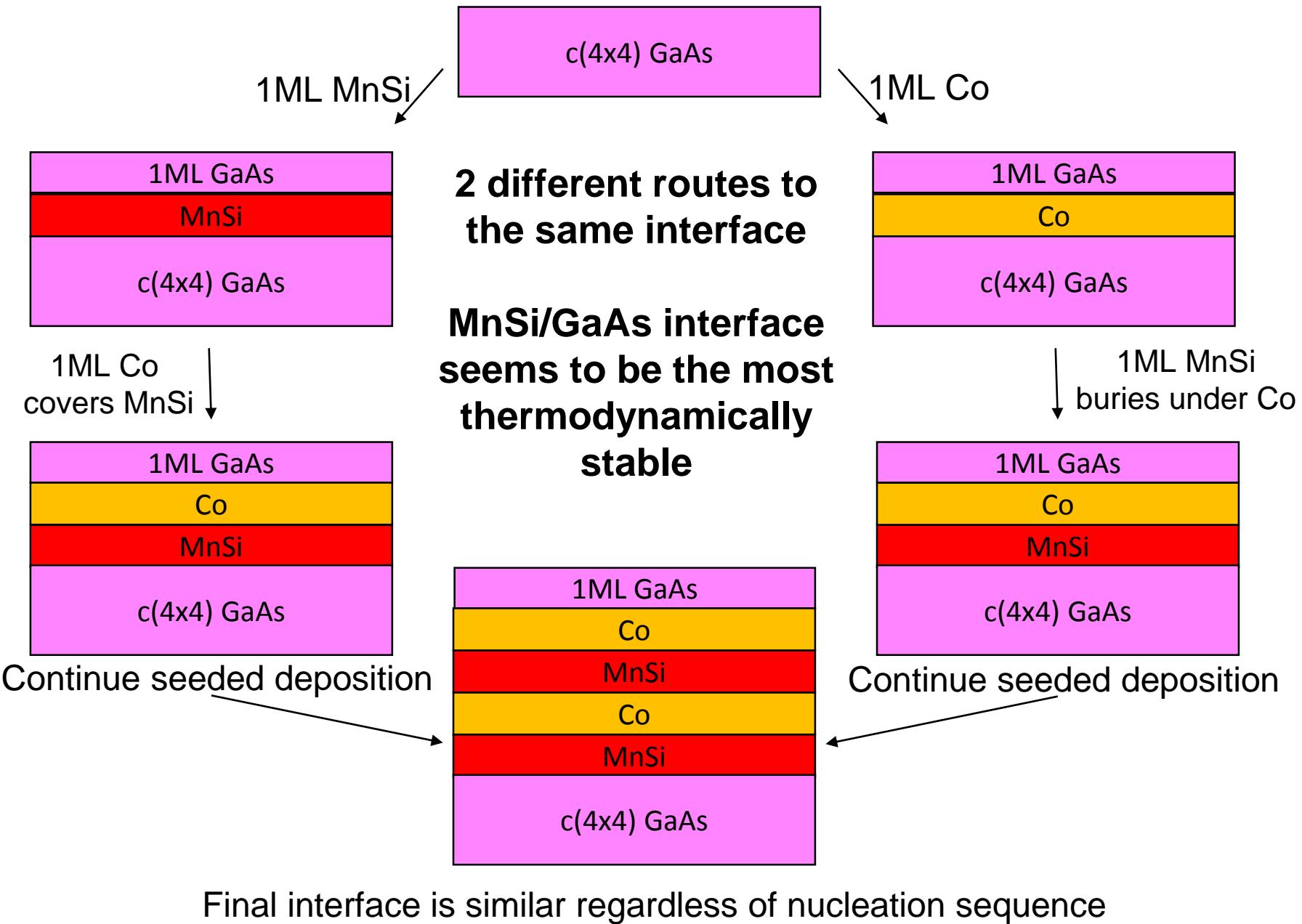
Co deposition attenuates Mn 2p peak

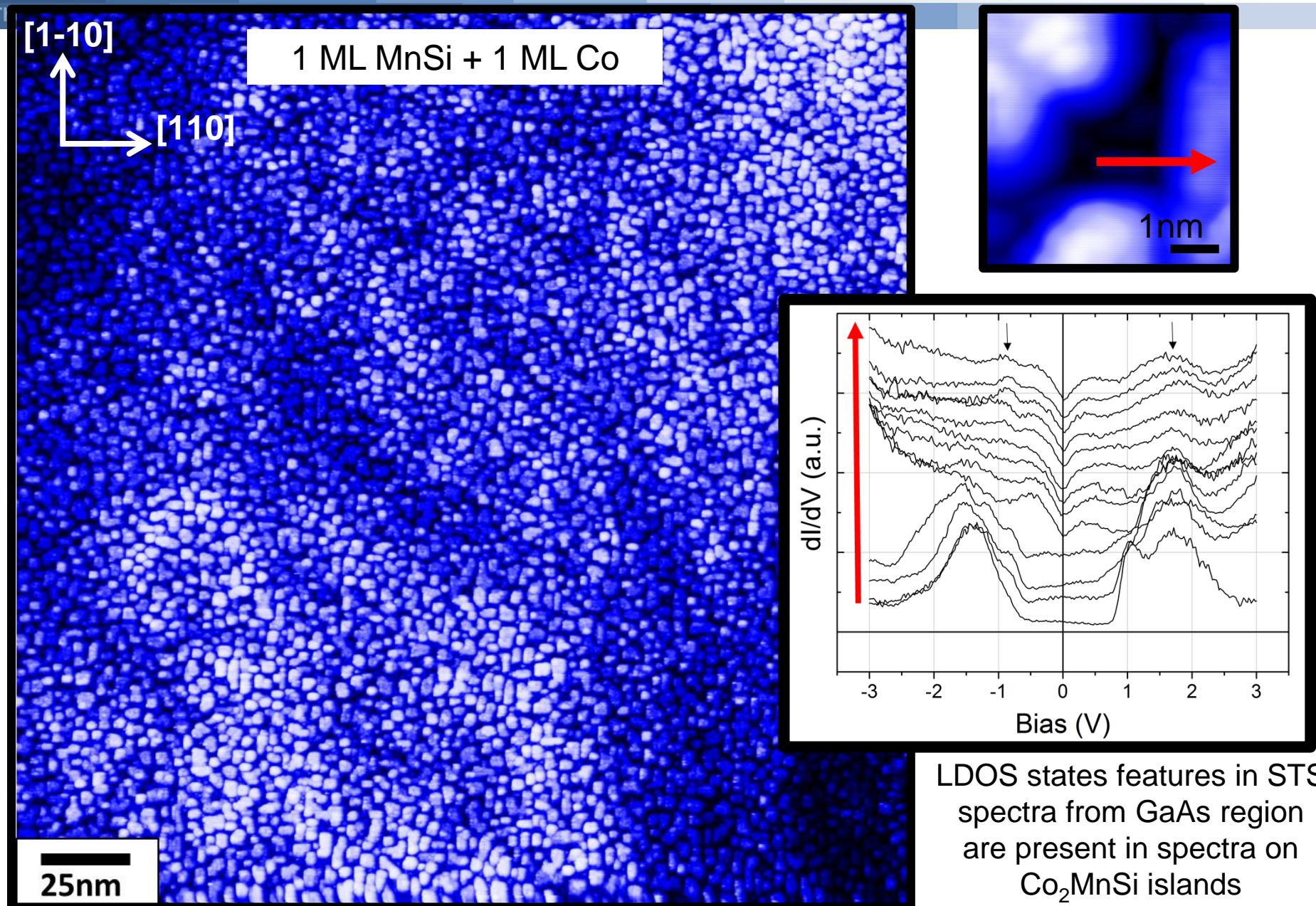
For first MnSi monolayer, the Co 2p peak is not attenuated at all, indicating that MnSi goes “under” the first Co layer

MnSi/GaAs interface is the most stable and forms regardless of deposition sequence

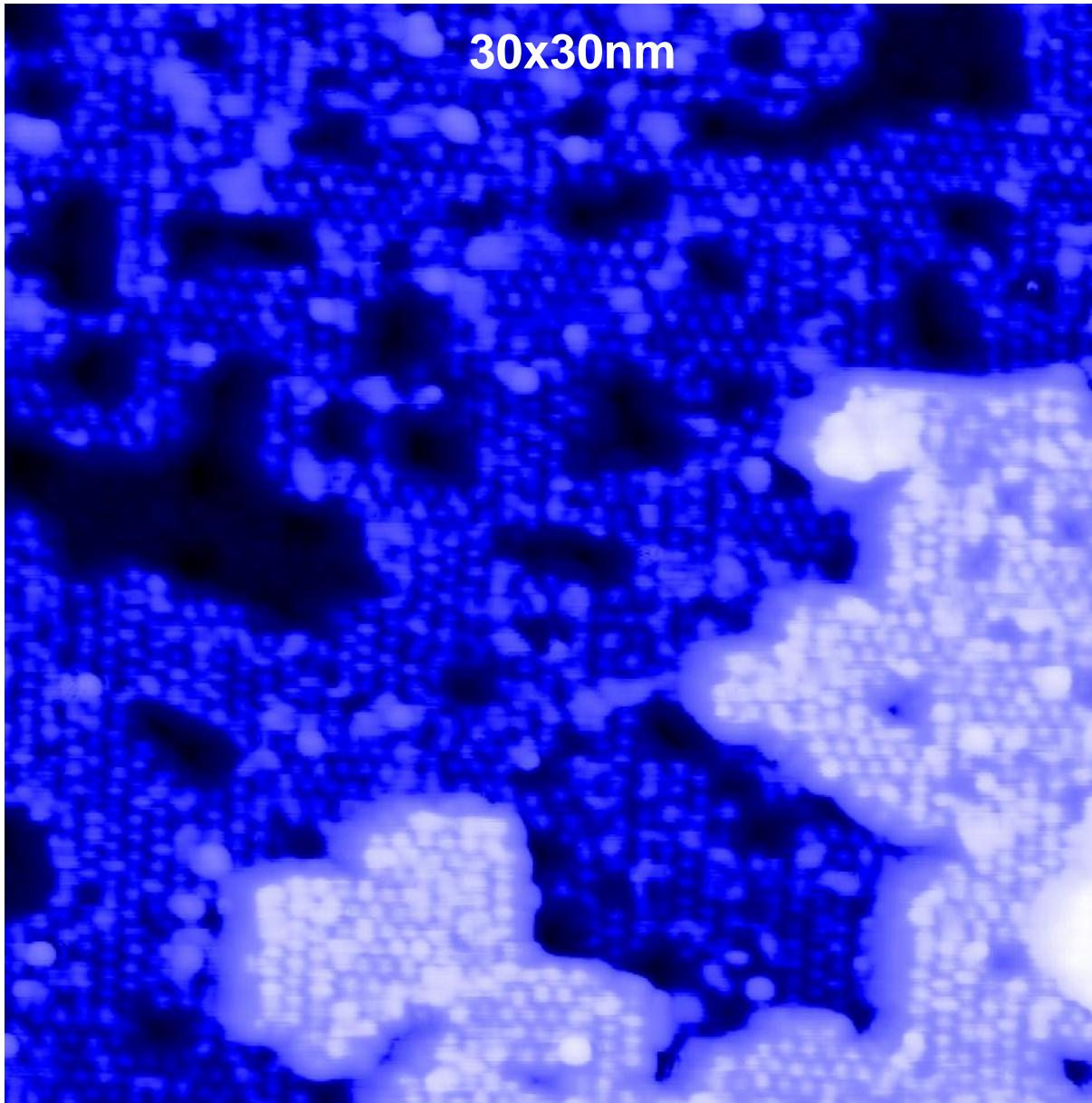


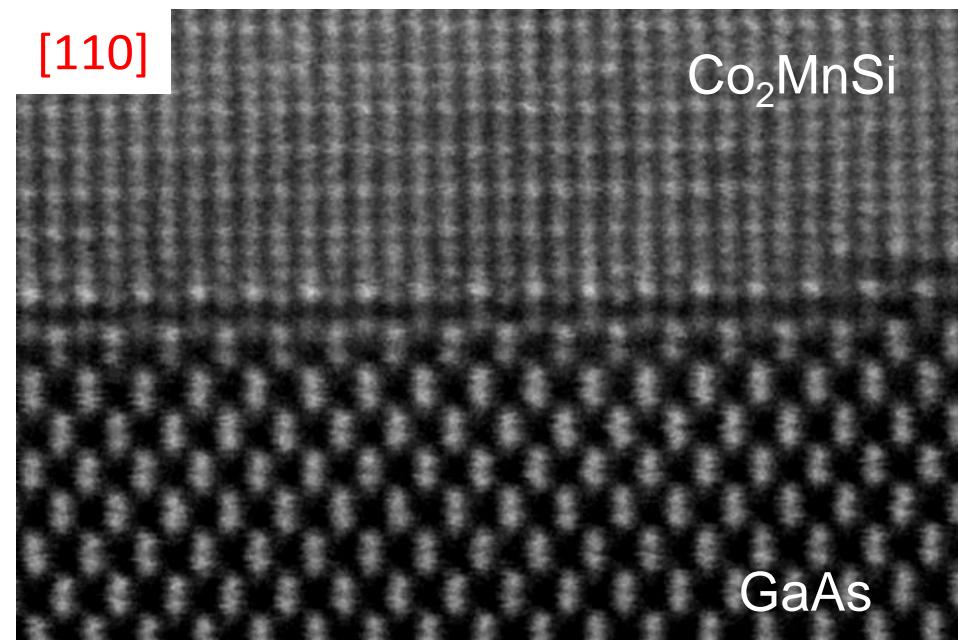
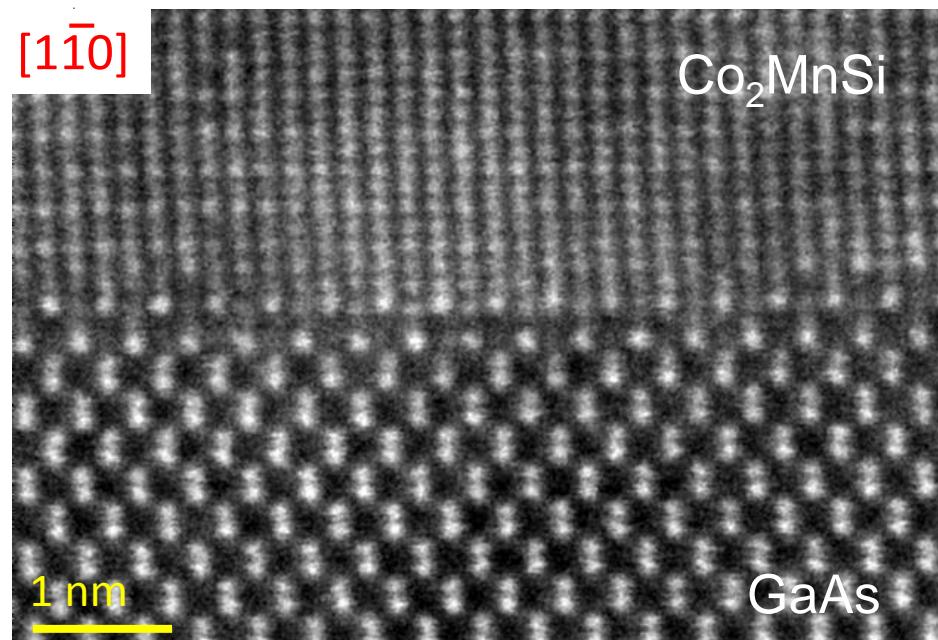
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LDOS states features in STS spectra from GaAs region are present in spectra on Co₂MnSi islands



T_g 270° C

- Co_2MnSi growth initiated by $\frac{1}{2}\text{ML MnSi}$

MnSi-initiated sample

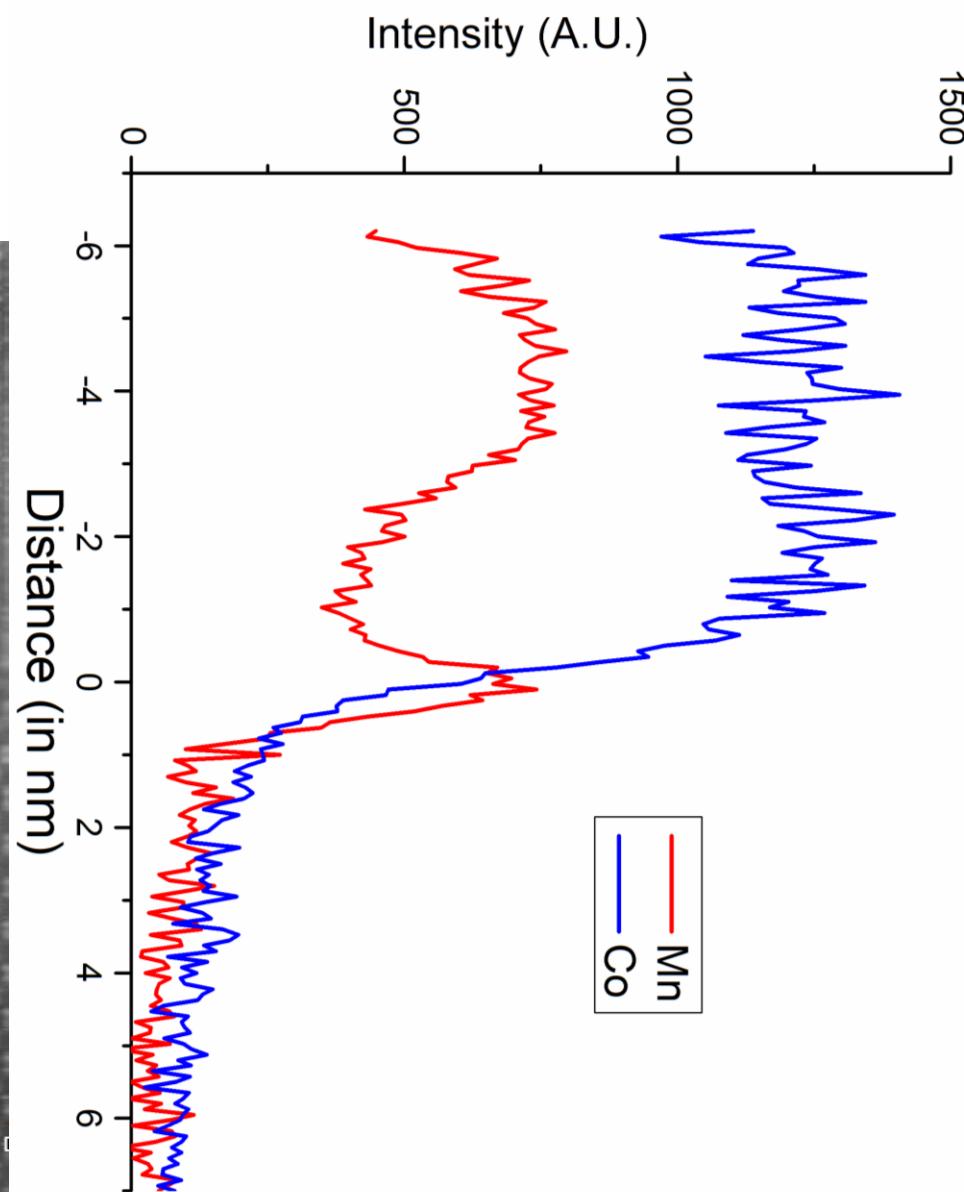
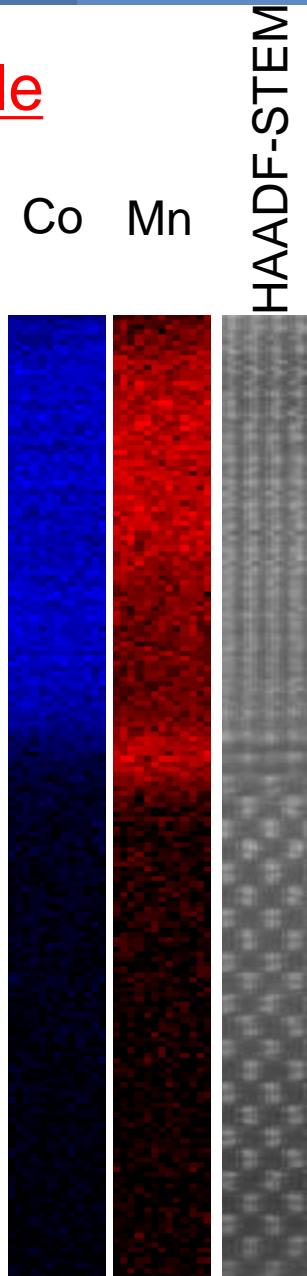
STEM-EELS mapping

shows the Mn-rich
interface

and diffusion of Mn

into the GaAs

Mn₂As-like formation at the
interface?

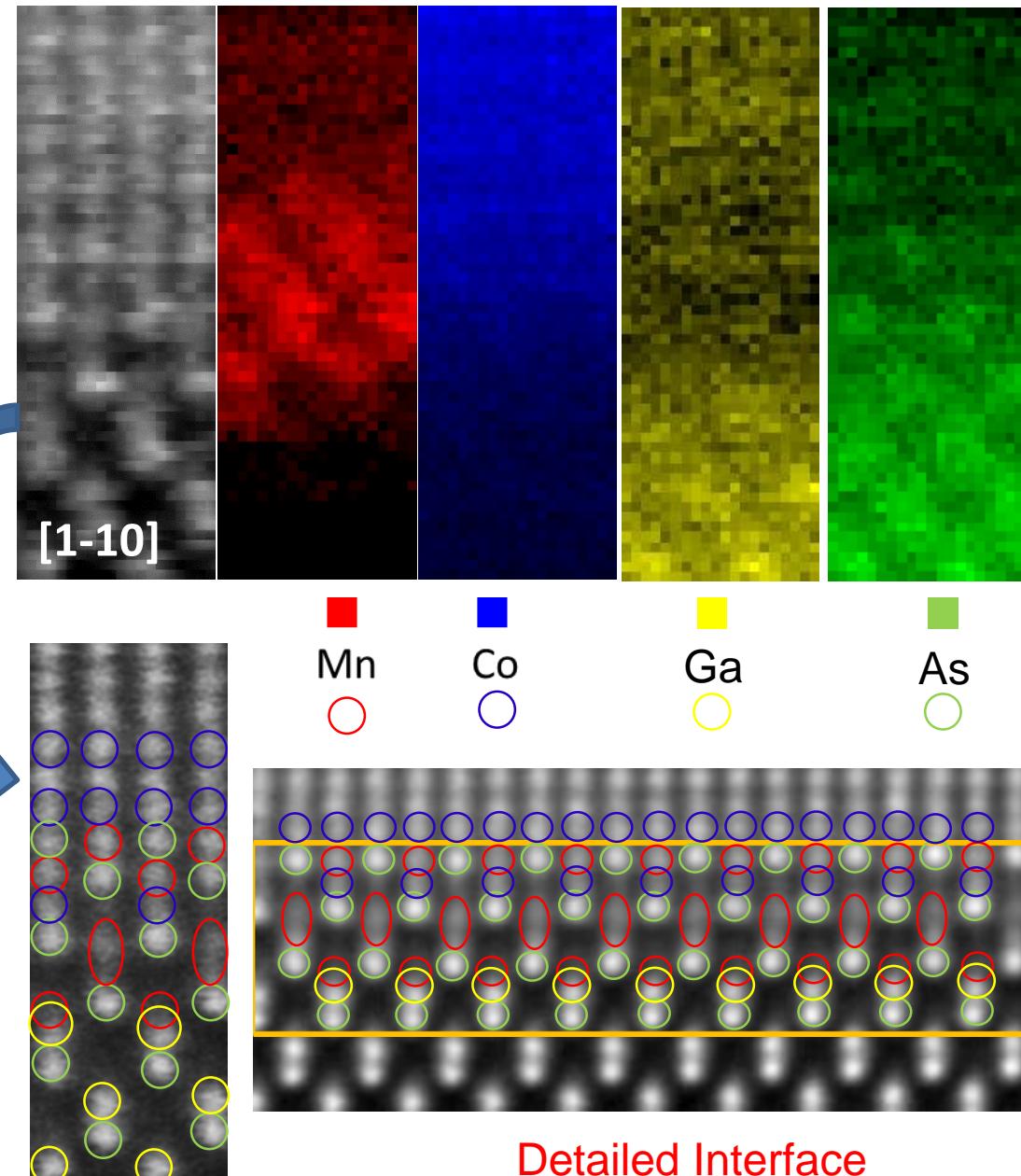


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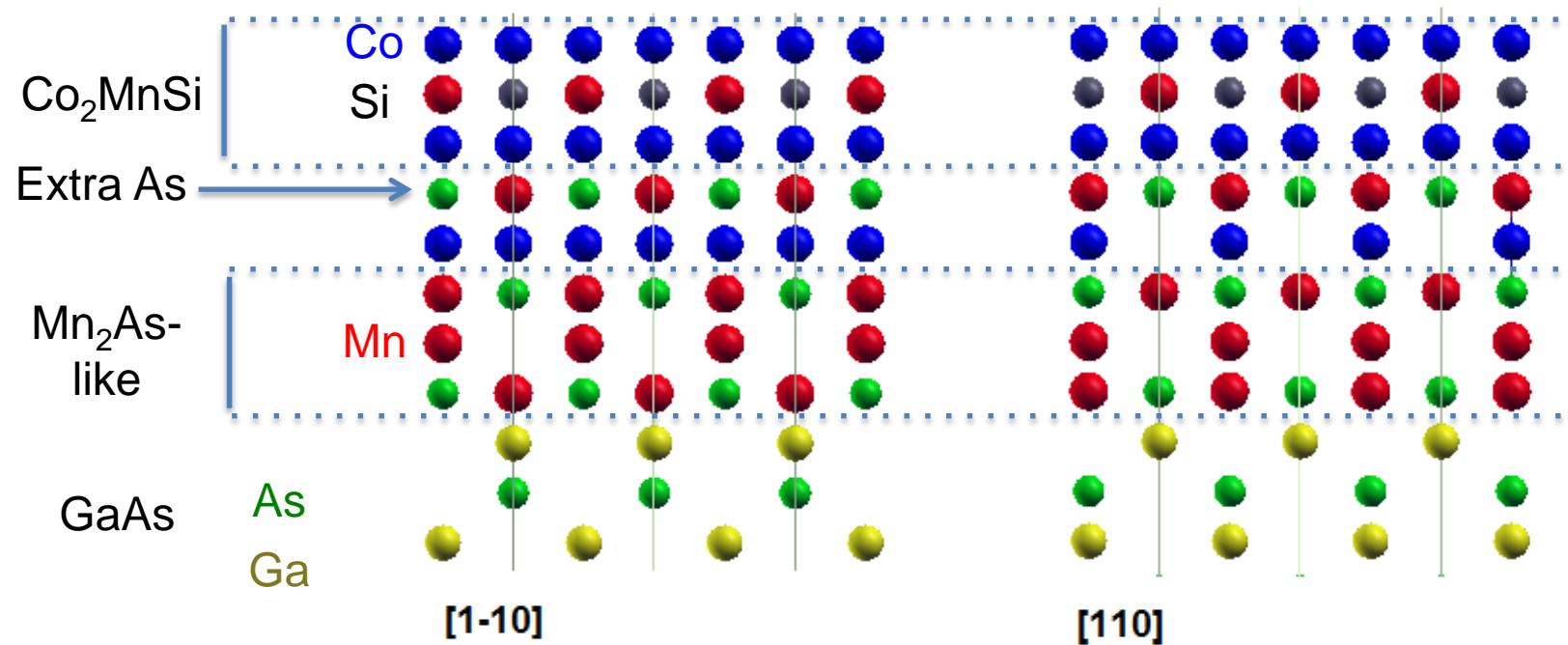
Mn atoms are six atomic layers away from the CMS layer and distributed inside the GaAs. Similarly, As atoms are

six planes away from GaAs and distributed inside CMS layer

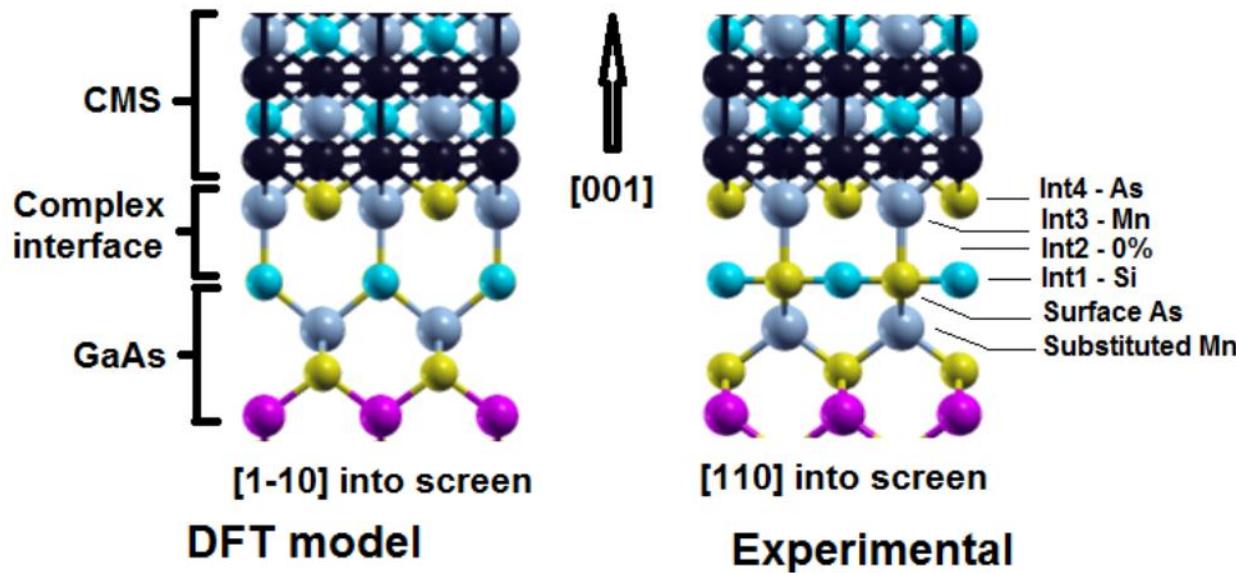
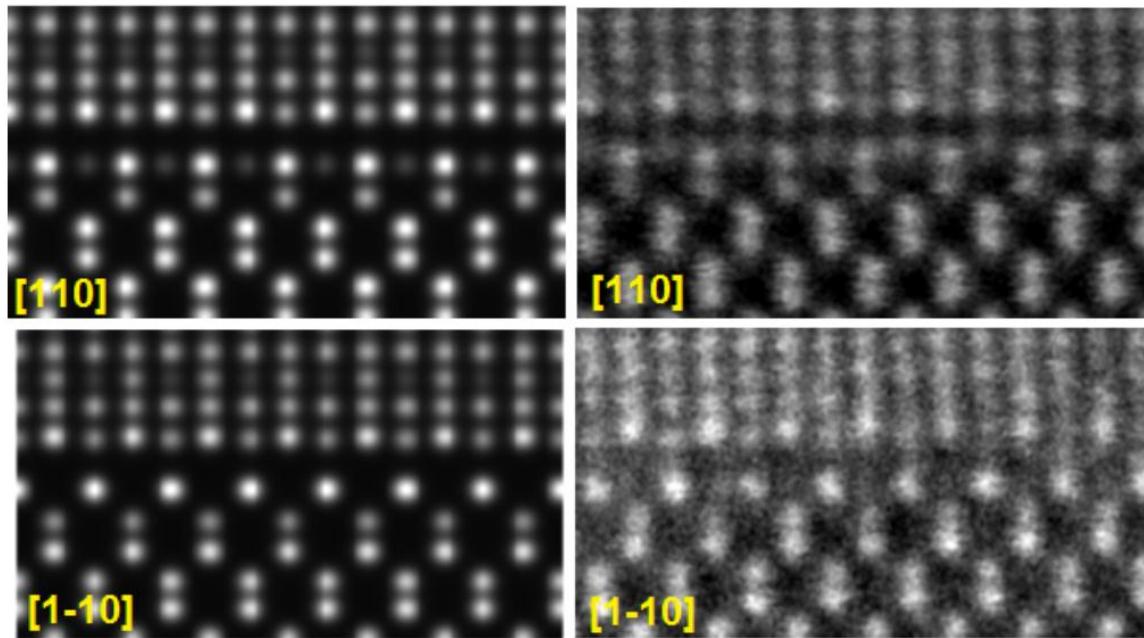
Unable to detect Si



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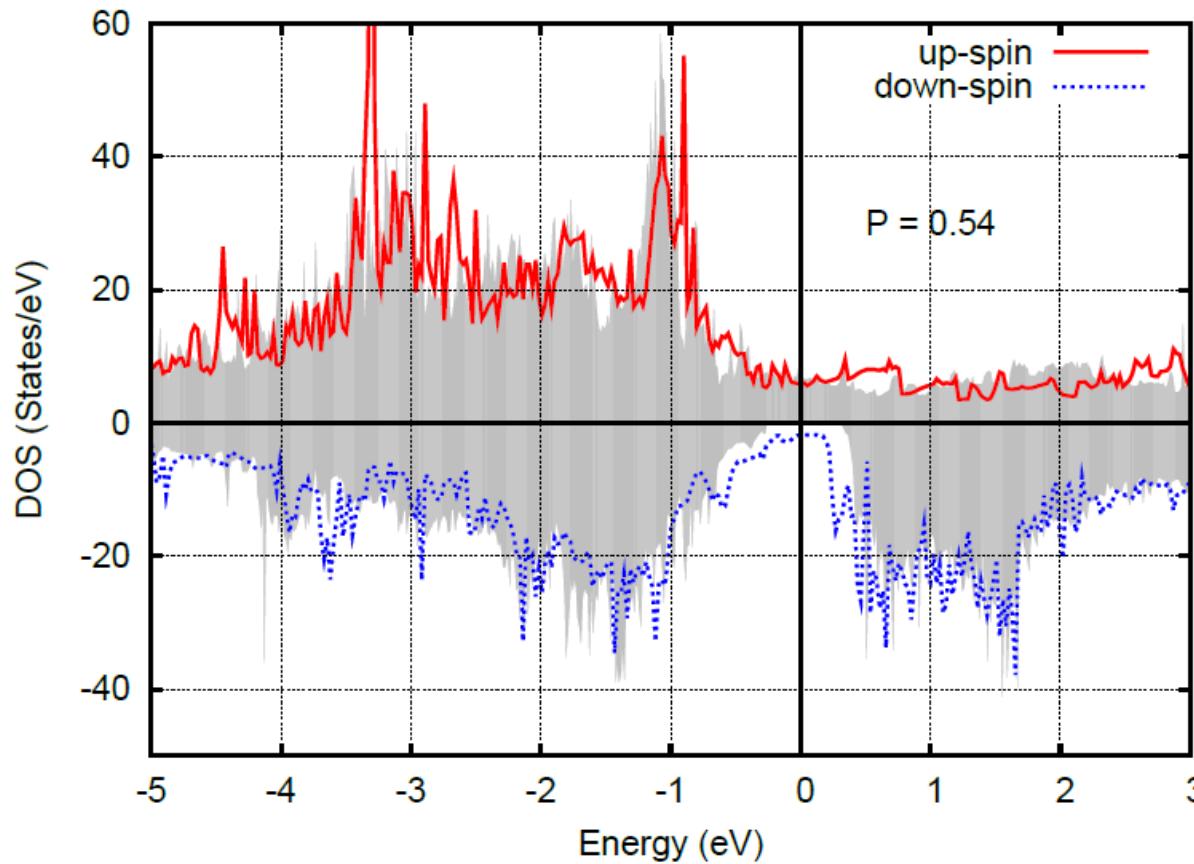


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Model #90C. Sivakumar
W. ButlerA. Rath
P. Voyles

Interface effect on the density states

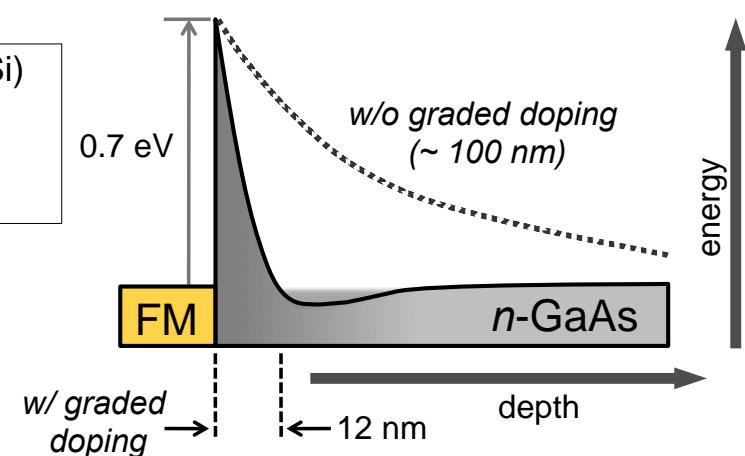
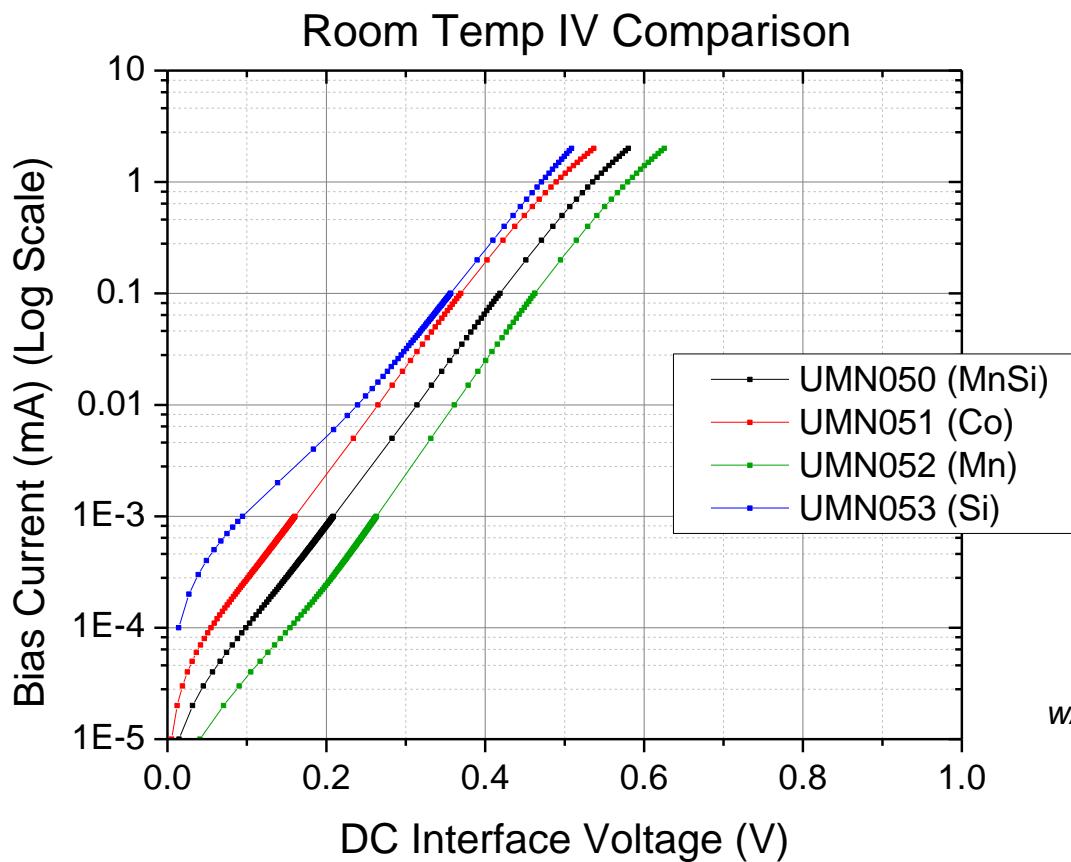
The convergence of research and innovation.



Red and blue Model #90 (best-fit model) DOS

Greyscale DOS is for an ideal abrupt termination of MnSi/As in CMS/GaAs (001)

Influence of initiation layer

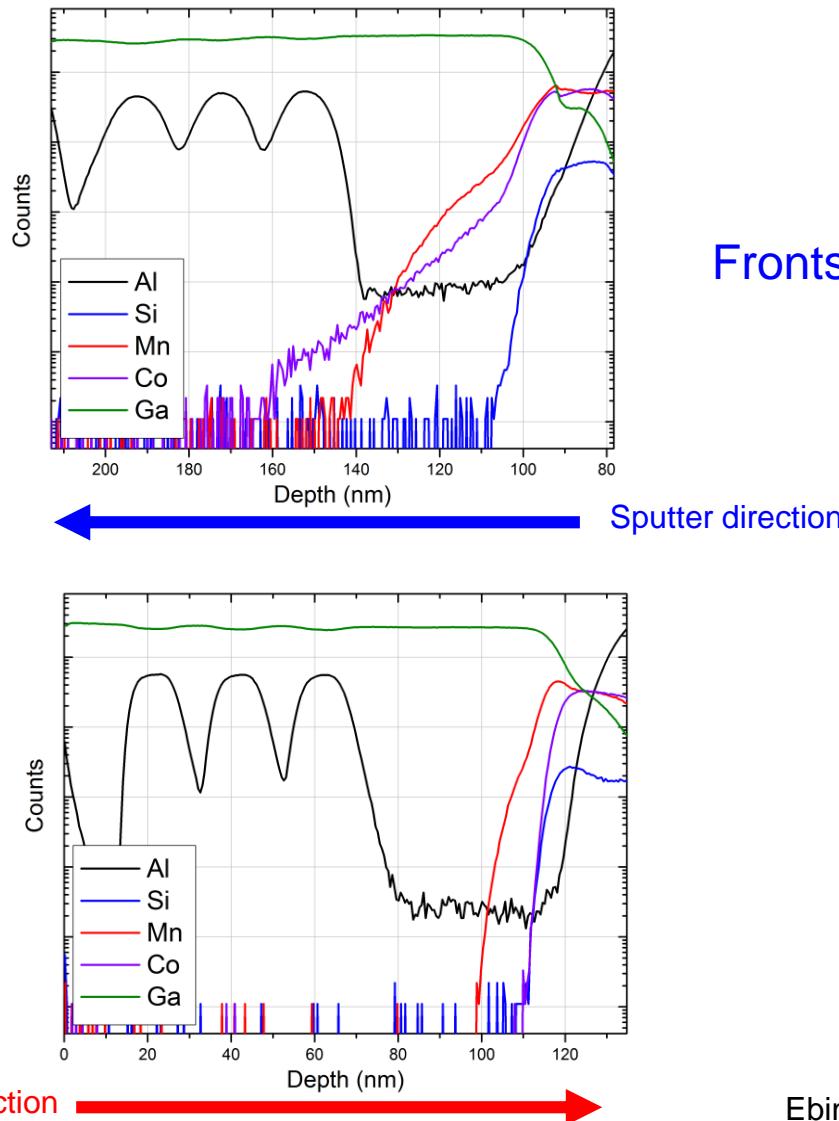


- Schottky barrier height change?
- Mn indiffusion?

The convergence of research and innovation.

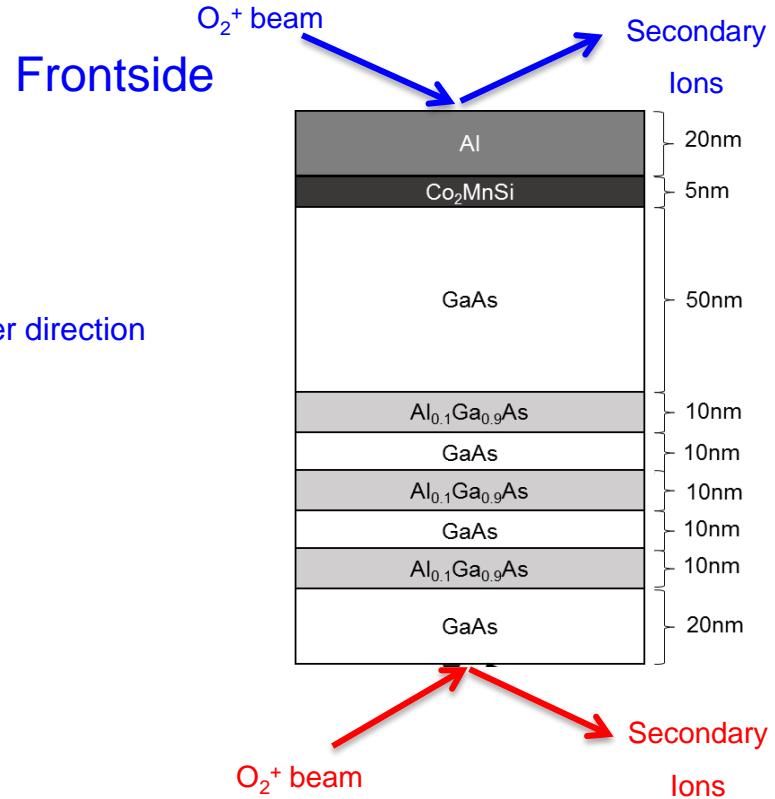
Is Mn compensating the n-type Shottky contact?

Frontside
(false profile)



Backside SIMS

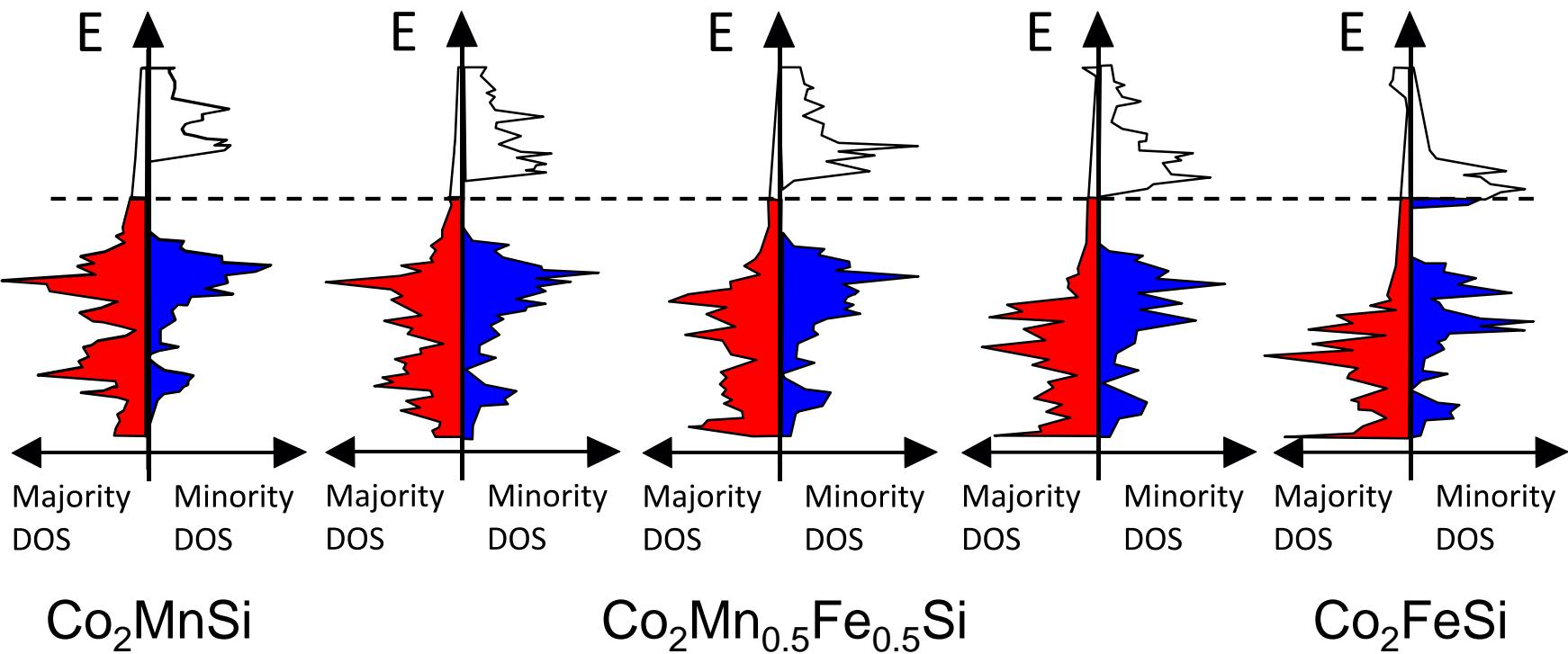
- Reduces knock-on of Co, Mn, and Si into GaAs



Ebina et al., Appl. Phys. Lett. **104**, 172405 (2014)

Tuning the Heusler Fermi Level

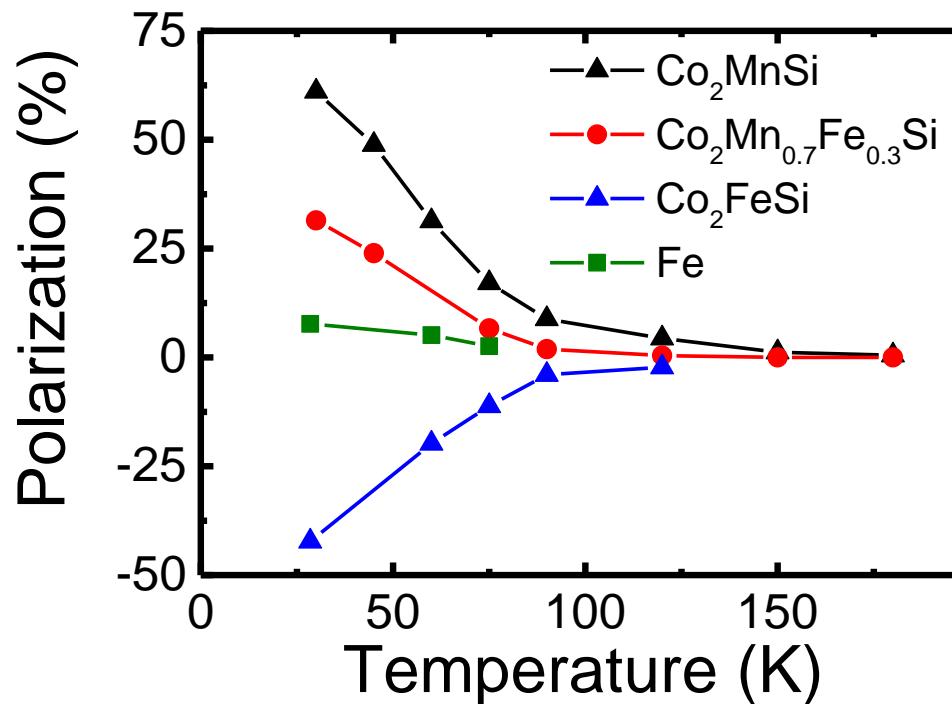
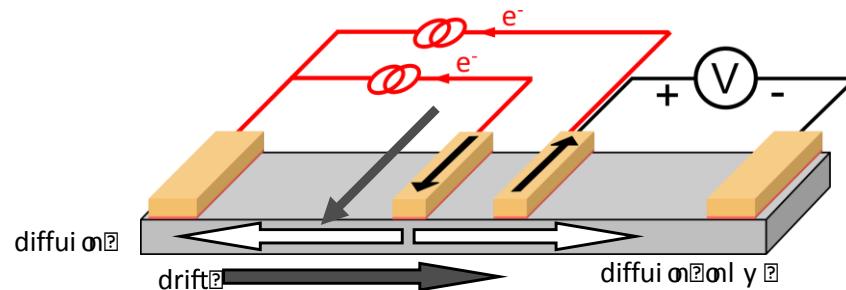
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- Co-doping Co_2MnSi with Fe increases the Fermi level
 - Co_2MnSi – 0.06% mismatch to GaAs
 - Co_2FeSi – 0.09% mismatch to GaAs

Co₂Mn_{1-x}Fe_xSi: comparison with Fe

The convergence of research and innovation.



- Polarizations determined by “biased detector technique”
- Sign change in going from Co₂MnSi to Co₂FeSi

Summary

- Demonstrated high quality MBE growth of Heusler compounds and integration with III-V semiconductors
- Demonstrated high spin polarization in GaAs
- Opposite sign of spin polarization for Co_2MnSi and Co_2FeSi
- Spin polarization can be tuned using $\text{Co}_2\text{Mn}_{1-x}\text{Fe}_x\text{Si}$
- Detailed interfacial structure is complicated – feedback between experiment and theory is essential for developing a consistent model of the interfacial atomic structure
- Strong evidence for Mn indiffusion into the GaAs